

Effects of Copper Substitution in Methylammonium-Based Perovskite Solar Cells [†]

Riku Okumura ¹, Takeo Oku ^{1*}, Atsushi Suzuki ¹, Sakiko Fukunishi ², Tomoharu Tachikawa ² and Tomoya Hasegawa ²

¹ Department of Materials Chemistry, The University of Shiga Prefecture, 2500 Hassaka, Hikone 522-8533, Japan; oe21rokumura@ec.usp.ac.jp (R.O.); suzuki@mat.usp.ac.jp (A.S.)

² Osaka Gas Chemicals Co., Ltd., 5-11-61 Torishima, Konohana-ku, Osaka 554-0051, Japan; fukunishi@ogc.co.jp (S.F.); t-tachikawa@ogc.co.jp (T.T.); hasegawa_tomoya@ogc.co.jp (T.H.)

* Correspondence: oku@mat.usp.ac.jp; Tel.: +81-749-28-8368

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Abstract: Addition of copper bromide to the perovskite precursor solutions increased the conversion efficiencies of the devices. On the other hand, the short-circuit current densities decreased with increasing the added amounts of copper (Cu). From first-principles calculations, partial substitution of lead with Cu resulted in the formation of a Cu d orbital energy level in the forbidden band, which worked as a recombination center, causing the generated carriers to disappear. Experiments and calculations show the effects of Cu substitution on the electronic structures and the ability of the addition of Cu compounds to further improve the device performance.

Keywords: copper; first-principles calculations; perovskite; solar cell

1. Introduction

Perovskite materials with excellent photovoltaic properties have been studied using a wide variety of approaches, including first-principles calculations [1–5], machine learning [6–10] and device characteristics simulations [11–15], in addition to experiments [16–22]. Although the conversion efficiency and stability of perovskite solar cells are gradually improving, most perovskite materials with excellent performance contain toxic Pb. In order to reduce toxicity for commercialization, alternative elements to Pb are being investigated [23–29]. In previous studies investigating the effect of Cu compound addition in methylammonium (MA) or Cs-based perovskites, it was reported that the additions of small amounts of Cu to the perovskite precursor solution increased the grain size and improved the film quality, which contributed to improved device properties [30–34]. Furthermore, the combination of Cu with alkali metals or organic cations more stable than MA improved the conversion efficiency and stability of perovskite solar cells [35–38].

In this study, the effects of the addition of Cu compounds to the perovskite precursor solution and the substitution of Pb with Cu on the device properties and electronic structure were investigated [39]. The amount of Cu added was varied in the range of 0, 1, 2, 3 and 12.5%, and current-voltage characterization and X-ray diffraction measurements were performed. In addition, first-principles calculations were performed to determine the effect of Cu substitution in methylammonium-based perovskite crystals from the band structure and partial density of states.

2. Device Fabrication and Computational Conditions

The structure of the fabricated perovskite solar cell is fluorine doped tin oxide glass/compact TiO₂/mesoporous TiO₂/CH₃NH₃PbI₃/spiro-OMeTAD/Au. CuBr₂ was used as the Cu compound to be added to the perovskite precursor solution, with 0% Cu as the

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standard device and different Cu additions in the range of 1, 2, 3 and 12.5%. The details of the experimental methods are described in the previous papers [35].

A $2 \times 2 \times 2$ supercell was built and one of the eight B-sites was substituted with Cu to produce a structural model of $\text{MAPb}_{0.875}\text{Cu}_{0.125}\text{I}_3$. The energy gap and carrier effective mass were calculated from the band structure, and information on orbitals was obtained from the partial density of states (pDOS). The details of the calculation method are described in the previous papers [39].

3. Results and Discussion

Figure 1 shows the cell parameters as a function of the amount of Cu compound added to the perovskite precursor solution. Table 1 shows the values of the cell parameters obtained from the current-voltage characterization. The highest open circuit voltage (V_{oc}) and fill factor (FF) were obtained when 2% Cu was added, and the conversion efficiency was higher than standard devices. The short-circuit current density (J_{sc}) tended to decrease with increasing Cu addition, and the conversion efficiency decreased compared to the standard device when the Cu content exceeded 2%.

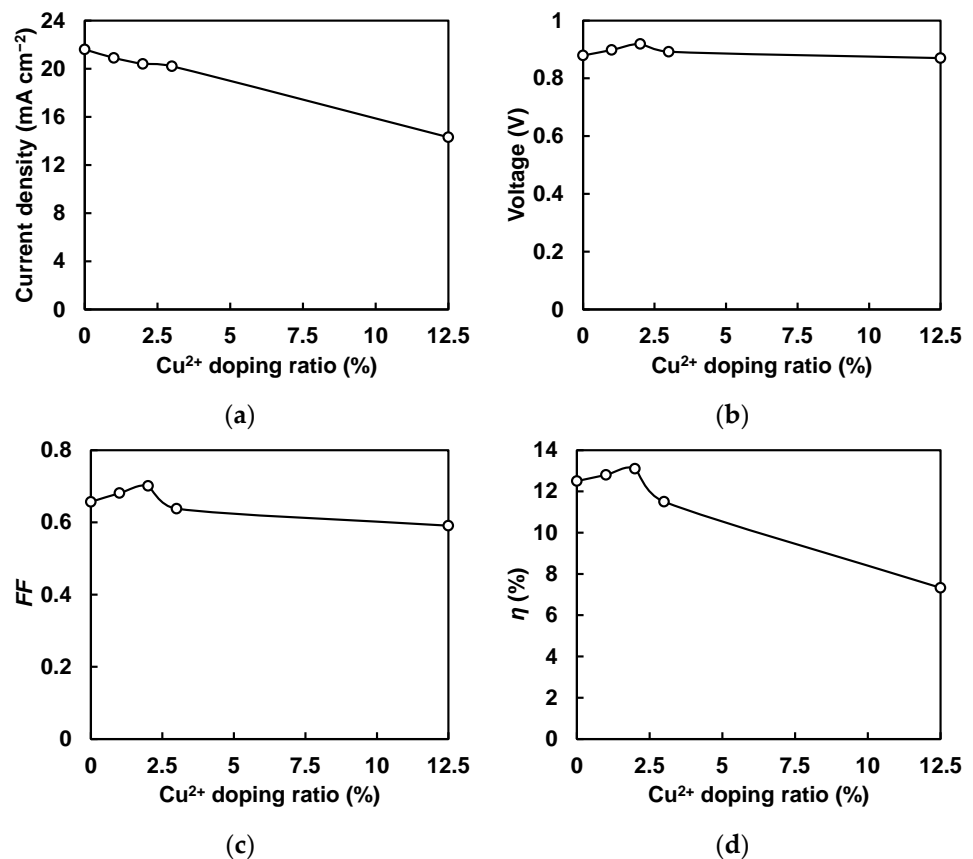


Figure 1. Cell parameters as a function of the amount of Cu compound added to the perovskite precursor solution.

Table 1. Device properties of perovskite solar cells.

Cu ²⁺ (%)	J_{sc} (mA cm ⁻²)	V_{oc} (V)	FF	R_s (Ω cm ²)	R_{sh} (Ω cm ²)	η (%)	η_{ave} (%)	E_g (eV)
0	21.6	0.879	0.657	4.04	1664	12.5	10.4	1.55
1	20.9	0.898	0.681	3.49	1526	12.8	11.6	1.55
2	20.4	0.919	0.701	3.03	1507	13.1	11.7	1.56
3	20.2	0.892	0.638	4.85	1213	11.5	10.5	1.56

12.5	14.3	0.870	0.591	5.07	621	7.33	6.07	1.60
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Table 2 shows the results of the X-ray diffraction pattern analysis. The crystallite size increased with the addition of Cu to the precursor solution, with 2% Cu showing the highest conversion efficiency and the highest perovskite (100) plane orientation. The increase in the lattice parameter with the addition of small amounts of Cu was attributed to lattice distortion due to Cu substitution. For 12.5% Cu, the ionic radius of Cu is smaller than that of Pb, so the lattice shrank as more Cu was incorporated into the perovskite crystals.

Table 2. Parameters obtained from X-ray diffraction measurements.

Cu ²⁺ (%)	I_{100}/I_{210}	Lattice Parameter (Å)	Crystallite Size (Å)
0	3.4	6.267(0)	394
1	3.7	6.269(1)	549
2	4.6	6.268(0)	663
3	4.2	6.268(1)	548
12.5	2.2	6.243(0)	625

First-principles calculations were performed to investigate the cause of the decrease in J_{sc} with increasing Cu addition. Figure 2 shows the calculated band structure and density of partial states. Table 3 shows the values of the parameters obtained from the first-principles calculations. From Figure 2c, the valence and conduction bands are composed of I p and Pb p orbitals, respectively. When Pb is partially substituted by Cu, the band gap decreases due to the lower energy of the Pb p orbital. Furthermore, an energy level of the Cu d orbital was formed in the forbidden band. Considering the experimental results, the energy level of the Cu d orbital worked as a defect level, causing the loss of the generated carriers and the reduction of the J_{sc} . In the absence of Cu, electrons are excited from the I p orbital to the Pb p orbital. In the presence of Cu, electrons can be excited from the I p orbital to the Cu d or Pb p orbital, resulting in two patterns of excitation processes. Despite the increase in the excitation probability, the decrease in device properties with increasing Cu addition means that it is difficult to extract electrons excited from the I p orbital and trapped in the Cu d orbital as charge carriers. Experimental and computational results indicate that Cu substitution has a negative effect on the electronic structure and that the defect levels formed in the forbidden bands reduce the J_{sc} . Therefore, the enhancement of device properties by the addition of Cu compounds to the precursor solution is attributed to improvements in the microstructure of the perovskite film, such as an increase in crystallite size and perovskite crystal (100) plane orientation.

Table 3. Parameters obtained from first-principles calculations.

Model	Total Energy (eV cell ⁻¹)	Energy Gap (eV)	m^*/m_0			Contribution of Transition (%)	Oscillator Strength
			Pb	I	Cu		
MAPbI ₃	-3495	1.480	0.229	0.208	–	16.7	0.0168
Cu 12.5%	-3363	1.438	0.240	0.239	0.355	62.2	0.0287

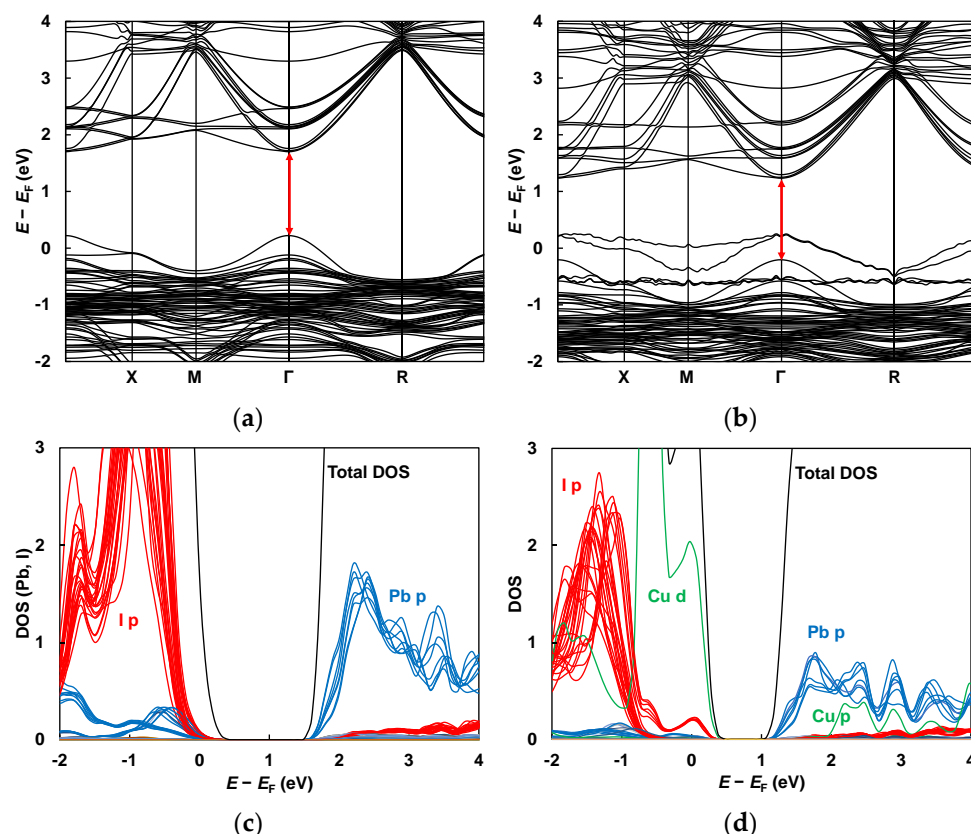


Figure 2. Calculated band structures and DOS of (a,c) MAPbI₃ and (b,d) MAPb_{0.875}Cu_{0.125}I₃.

4. Conclusions

The addition of 2% Cu to the perovskite precursor solution improved the V_{oc} and FF and enhanced the conversion efficiency of the device. The calculated band structure and density of partial states indicate that the substitution of Pb for Cu has a negative effect on the electronic structure, with the formation of Cu d-orbital energy levels in the forbidden band causing carrier recombination and lowering the J_{sc} . Therefore, it is difficult to adopt Cu as a replacement element for Pb, but the addition of small amounts of Cu compounds to the perovskite precursor solution was shown to contribute to further enhancement of device properties.

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