

Lithium niobate: from single crystals to nanocrystals

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Abstract: LiNbO₃ single crystals were first grown more than 50 years ago. Since that time thousands of papers have been published dealing with their outstanding ferroelectric, acoustic, nonlinear optical, holographic etc. properties and demonstrating their countless realized or potential applications. It was about 25 years ago, when the first stoichiometric LiNbO₃ crystals gave a new impulse to the never-ending investigations. Different applications require different undoped or doped systems of bulk, thin film, or nanocrystal forms. In the present talk I'll show two examples, (i) incorporation of dopants into stoichiometric crystals, and (ii) properties of LiNbO₃ nanocrystals prepared by high-energy ball-milling.

Dopants are generally used to tailor the crystal properties for a given application. To understand the effect of dopants the substitution site in the crystal have to be known. Our IR absorption studies unambiguously showed that for the di-, tri-, and tetravalent cations a threshold concentration exists above which the dopants partially substitute at Nb sites, while below it they can be found on Li sites.

Nano-crystalline LiNbO₃ was prepared from single crystals by the high-energy ball-milling technique. During milling the material suffered partial reduction that lead to the formation of bipolarons and polarons yielding gray color together with Li₂O segregation on the open surfaces. Upon high temperature oxidation a LiNb₃O₈ shell was formed. The particle size of the nano-crystals were determined by dynamic light scattering (DLS) and scanning electron microscopic (SEM) methods.

Keywords: lithium niobate; single crystal; nanocrystal;

OUTLINE

i) Dopant sites in stoichiometric LiNbO_3 single crystals

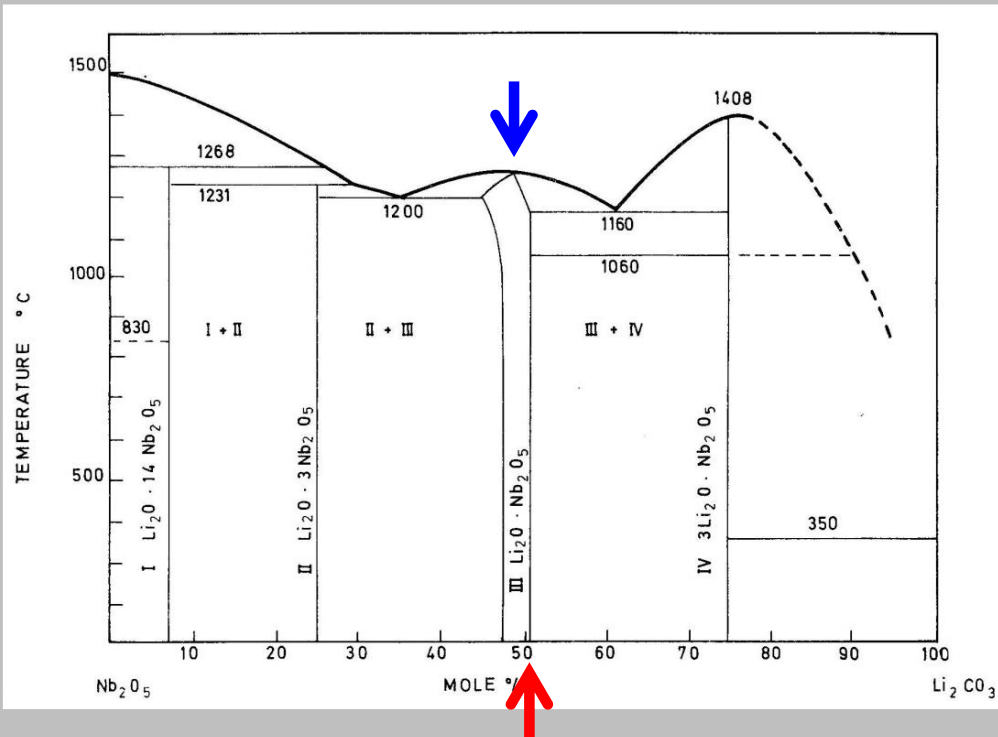
- Stoichiometric vs. congruent LiNbO_3 (SLN \leftrightarrow CLN)
- Hydroxyl ions in SLN
- Optical damage resistant (ODR) ions
- Rare-earth (RE) ions
- Transition metal (TM) ions (Fe^{3+} , Cr^{3+} , Ti^{4+})

ii) Mechanochemical reactions of LiNbO_3 induced by high-energy ball-milling

- Lithium niobate nanocrystals
- High-energy ball-milling (dry and wet grinding)
- Particle and grain size reduction
- Phase transformation and chemical reaction
- Structure characterization (X-ray, Raman, reflection spectroscopy, coulometric titration, electron microscopy)

i) Stoichiometric vs. congruent LiNbO_3

Phase diagram, crystal growth



LiNbO_3 melts congruently
grown by the Czochralski method

48.4 mol% Li_2O
51.6 mol% Nb_2O_5
 $\text{Li}/\text{Nb} \approx 0.94$
 $\text{Li}_{1-5z}\text{Nb}_{1+z}\text{O}_3$ ($z \approx 0.01$)

Stoichiometric crystals
grown by the HTTSSG method
from K_2O flux

$\text{K}/\text{Nb} = 0.31$, $\text{Li}/\text{Nb} = 1$ in the flux
 $\text{K} \approx 0$, $\text{Li}/\text{Nb} \approx 1$ in the crystal
 LiNbO_3 ($z \approx 0$)

K. Polgár, Á. Péter, L. Kovács, G. Corradi, Zs. Szaller: Growth of stoichiometric LiNbO_3 single crystals by top seeded solution growth method, *Journal of Crystal Growth*, **177**, 211-216, 1997.

i) Stoichiometric vs. congruent LiNbO_3

OH^- absorption band

OH^- ions are probes of the defect structure in LiNbO_3

Congruent ($\text{Li}/\text{Nb} \approx 0.94$)

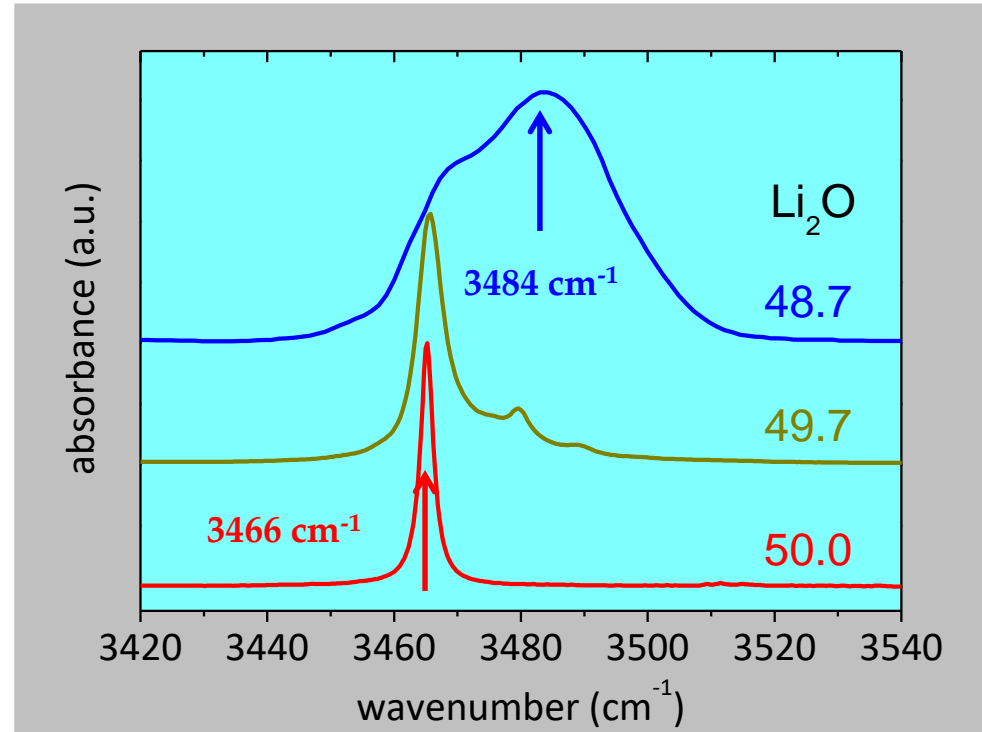
$\text{Li}_{1-5z}\text{Nb}_{1+z}\text{O}_3$ ($z \approx 0.01$)

$\nu_{\text{max}} \approx 3484 \text{ cm}^{-1}$

Stoichiometric ($\text{Li}/\text{Nb} \approx 1$)

LiNbO_3 ($z \approx 0$)

$\nu_{\text{max}} = 3466 \text{ cm}^{-1}$



Stretching vibration of OH^- ions in LiNbO_3

i) Dopant ions in LiNbO_3

- Transition metal (TM) ions: **$\text{Fe}^{2+/3+}$, Mn^{2+} , Cu^{+2+} , Ni^{2+} , Cr^{3+} , Ti^{4+} , etc.**
 - Increase the photorefractive sensitivity utilized in holographic recording
 - Surface layer diffused Ti^{4+} is used in optical waveguides
- Optical damage resistant (ODR) ions: **Mg^{2+} , Zn^{2+} , Sc^{3+} , In^{3+} , Hf^{4+} , Zr^{4+} , Sn^{4+}**
 - ODR ions above a **threshold concentration** suppress the photorefractive damage
 - The threshold concentration depends on the valence state of the dopant and the composition of LN (much lower for stoichiometric LN).
- Rare earth ions: **Pr^{3+} , Nd^{3+} , Dy^{3+} , Er^{3+} , Yb^{3+} , ...**
 - Laser active dopants, 4f - 4f transitions
 - Quantum Information Processing (QIP), Quantum Optics

i) Threshold concentration of ODR dopants

Below threshold $M^{n+} \rightarrow Li^+$ for $n = 2, 3, 4$ reducing the number of Nb_{Li}

Above threshold $M^{n+} \rightarrow Li^+$ and Nb^{5+} no Nb_{Li} ions are left in the crystal

Congruent $LiNbO_3$ ($Li_{1-5z}Nb_{1+z}O_3$ ($z \approx 0.01$), $Nb_{Li} \approx 0.01$)

Stoichiometric $LiNbO_3$ ($Li_{1-5z}Nb_{1+z}O_3$ ($z \approx 0$), $Nb_{Li} \approx 0$)

No excess Nb – no Nb_{Li} $C_{th} \approx 0$ mol%

Experimentally – almost fulfilled

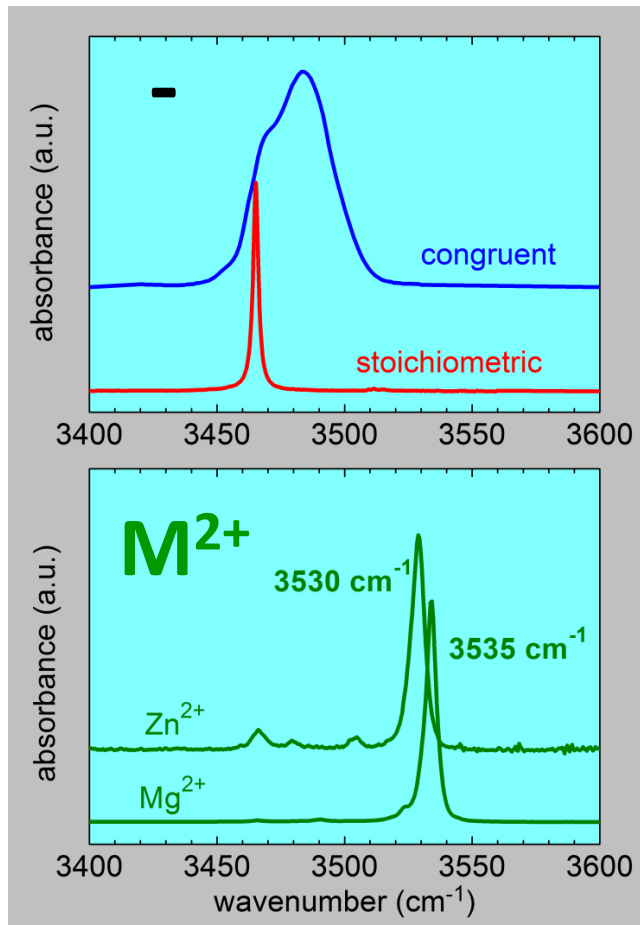
- for Mg^{2+} $C_{th} \approx 0.2$ mol% in sLN $\ll C_{th} \approx 5$ mol% in cLN
- for Zr^{4+} $C_{th} \approx 0.09$ mol% in sLN $\ll C_{th} \approx 2$ mol% in cLN

The OH^- absorption spectrum is one of the best indicator of the threshold effect

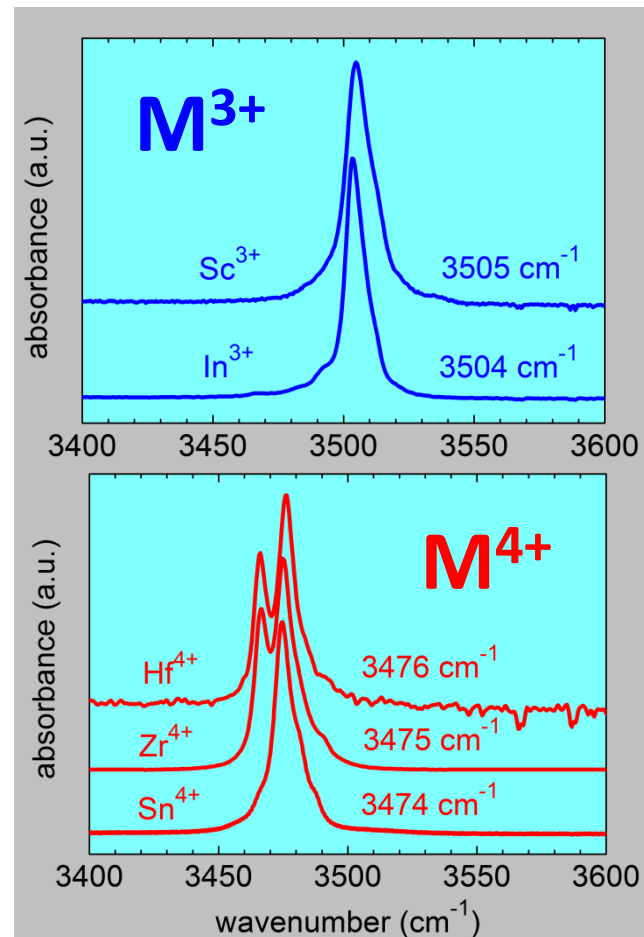
Á. Péter, K. Polgár, L. Kovács, K. Lengyel: Threshold concentration of MgO in near-stoichiometric $LiNbO_3$ crystals, *Journal of Crystal Growth*, **284**, 149-155, 2005.

L. Kovács, Zs. Szaller, K. Lengyel, Á. Péter, I. Hajdara, G. Mandula, L. Pálfalvi, J. Hebling: Photorefractive damage resistance threshold in stoichiometric $LiNbO_3:Zr$ crystals, *Optics Letters*, **38**, 2861-2864, 2013.

i) Hydroxyl ions in ODR ion doped SLN

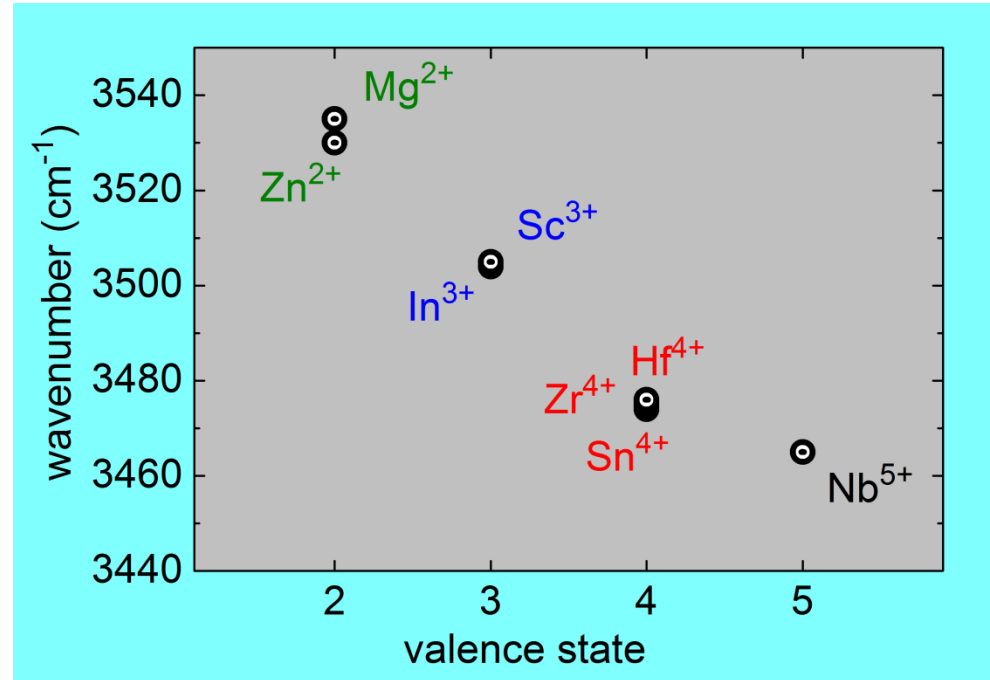
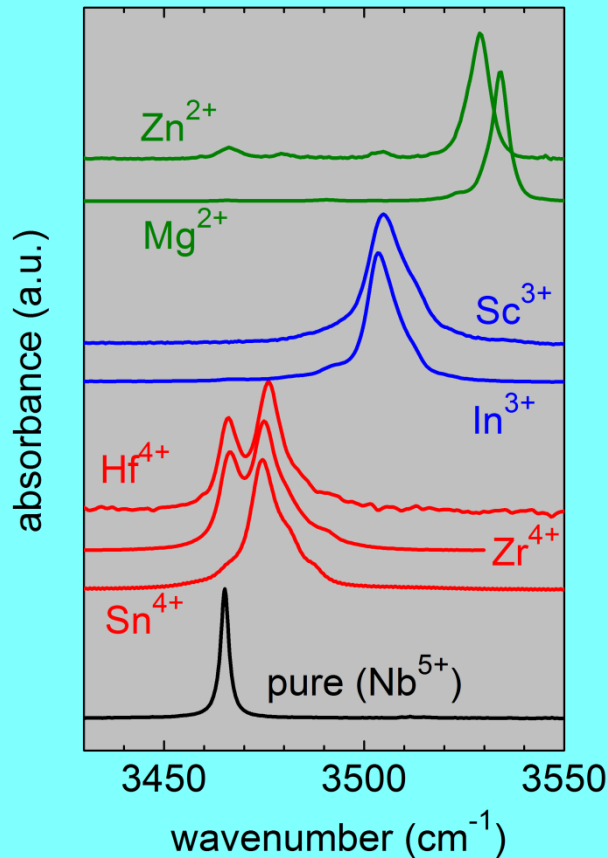


ODR ion	OH ⁻ vibrational frequency (cm^{-1})	
	below threshold	above threshold
Mg^{2+}	3466	3535
Zn^{2+}	3466	3530
In^{3+}	3466	3504
Sc^{3+}	3466	3505
Hf^{4+}	3466	3476
Zr^{4+}	3466	3475
Sn^{4+}	3466	3474



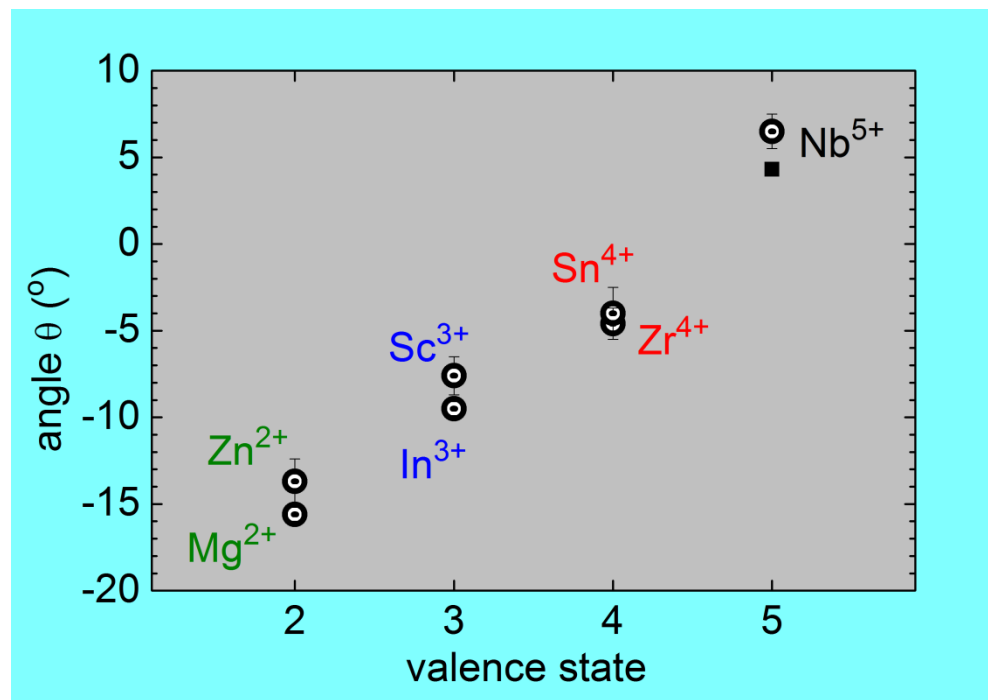
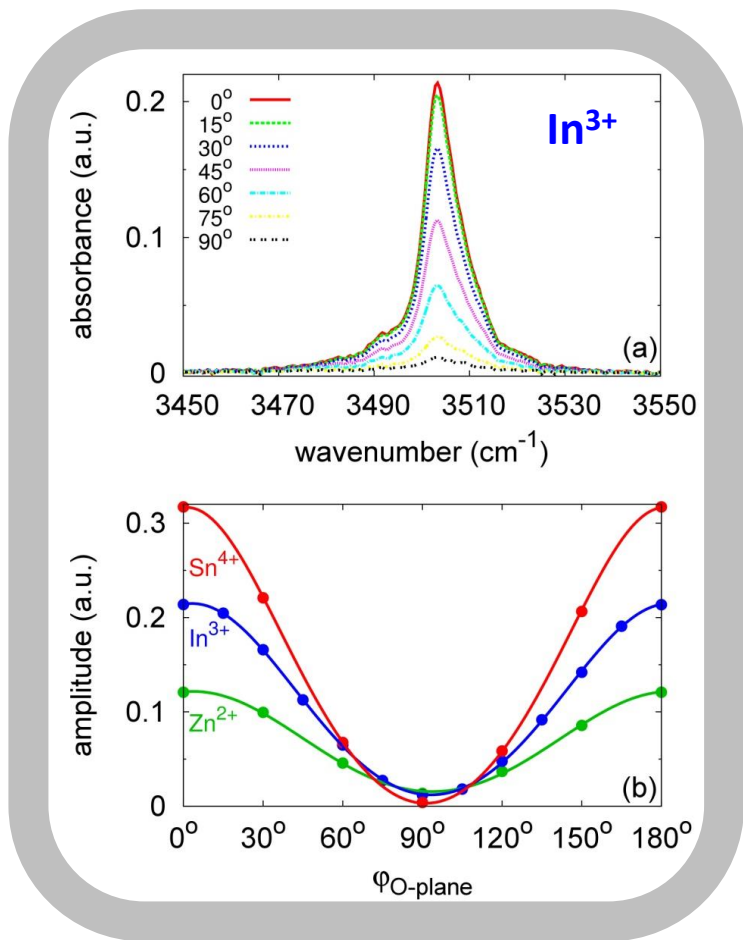
L. Kovács, Zs. Szaller, K. Lengyel, G. Corradi: Hydroxyl ions in stoichiometric LiNbO_3 crystals doped with optical damage resistant ions, *Optical Materials*, 37, 55-58, 2014.

i) Hydroxyl ions in ODR ion doped SLN



Stronger attractive forces for the protons in $\text{M}_{\text{Nb}}^{2+} - \text{OH}^-$ than in $\text{M}_{\text{Nb}}^{5+} - \text{OH}^-$

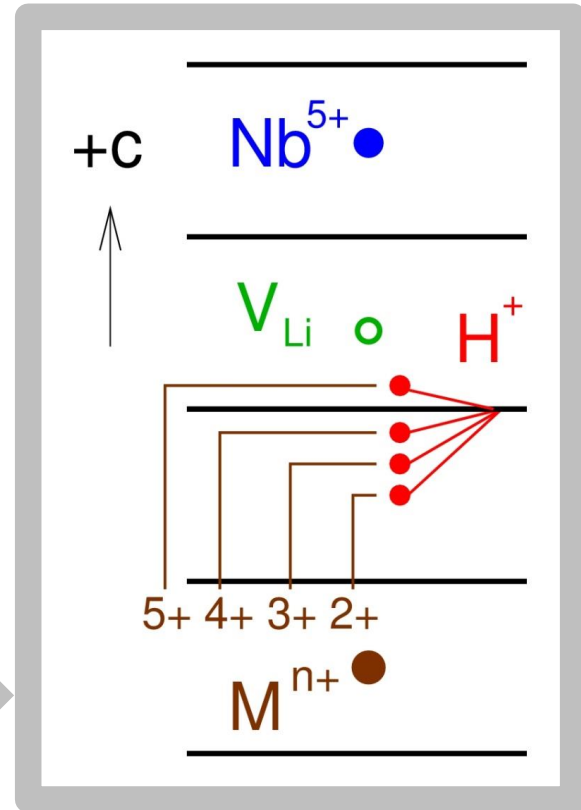
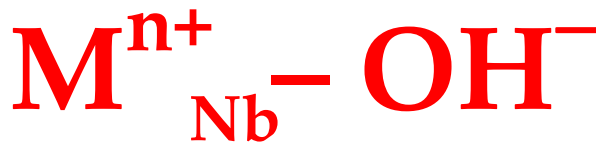
i) Hydroxyl ions in ODR ion doped SLN



θ is the angle between the O–H bond and the oxygen plane \perp to the c axis

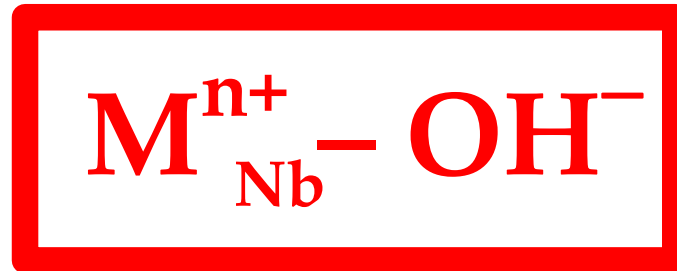
i) Hydroxyl ions in ODR ion doped SLN

- The new OH^- band is present if $C_{\text{M}^{n+}}$ is above the threshold
- OH^- is close to M^{n+}
- Above threshold M^{n+} at least partially occupies Nb site



i) Dopants in stoichiometric LiNbO_3

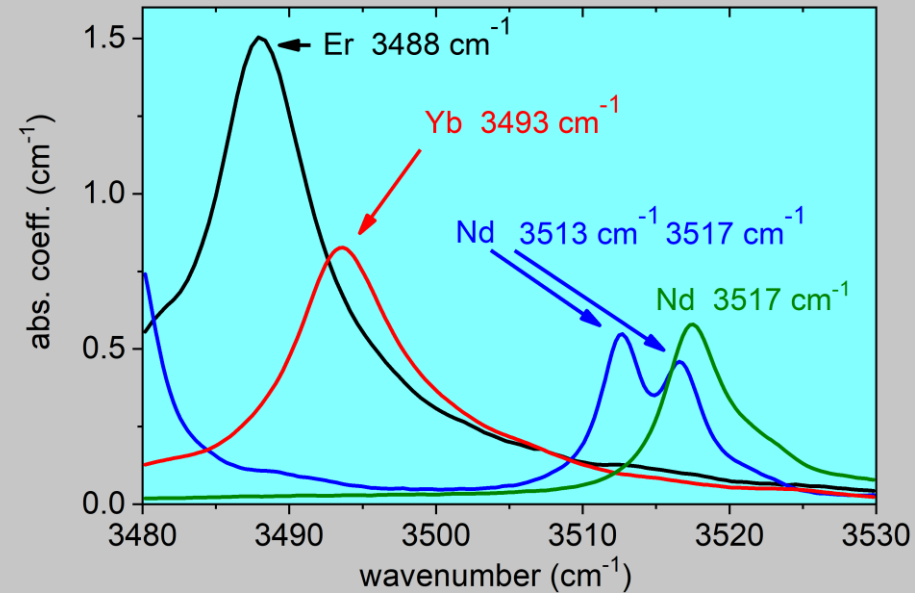
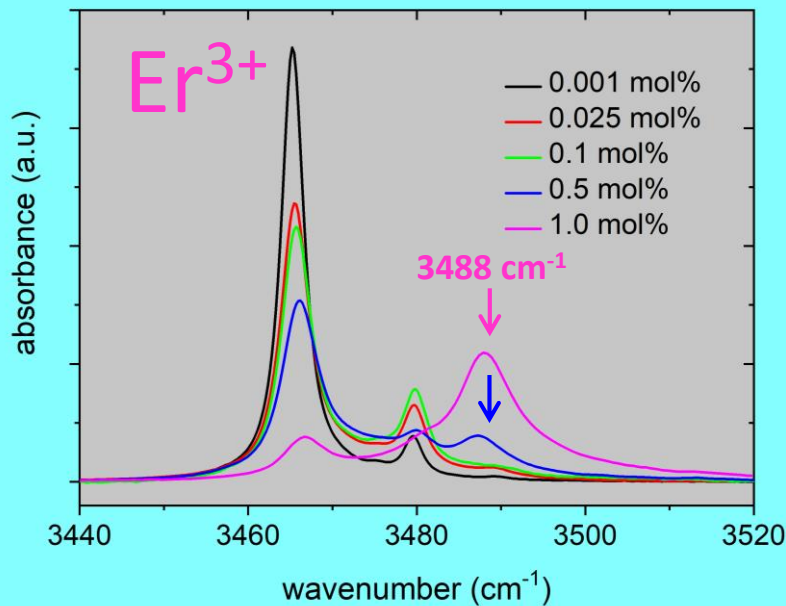
The defect model



based on the „threshold effect“ is valid for all ODR ions.

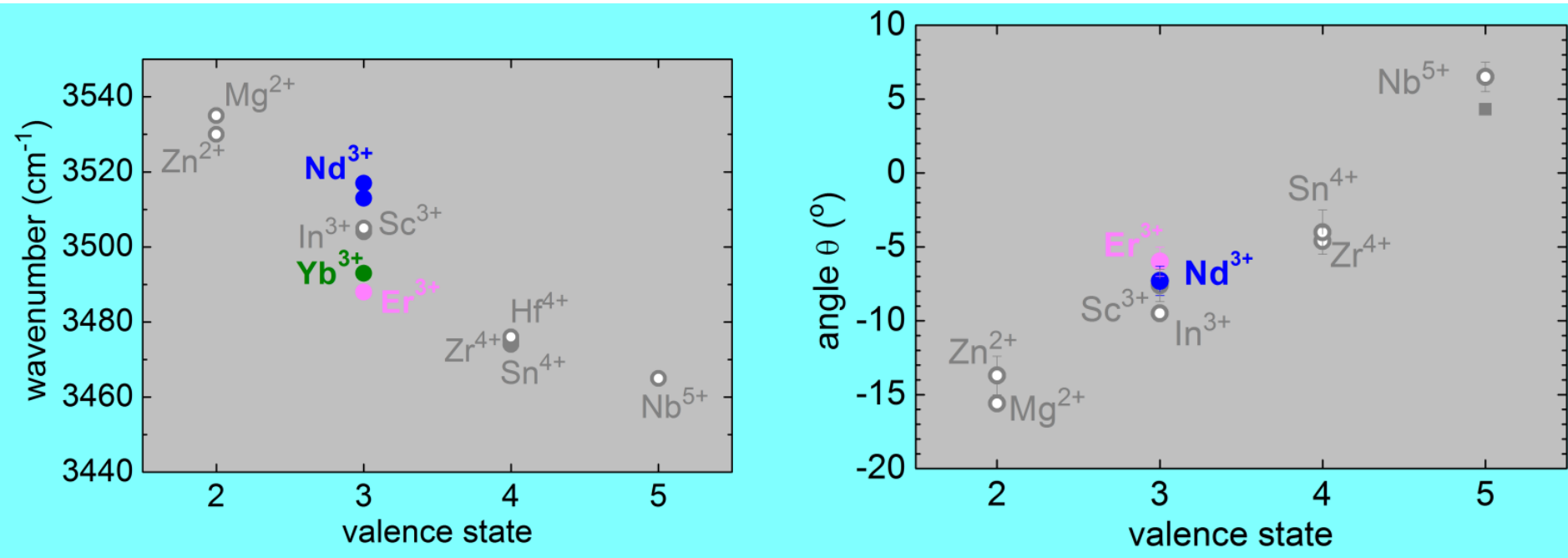
It should also work for $\text{M}^{\text{n}+} = \text{RE}^{\text{n}+}$ and $\text{TM}^{\text{n}+}$ ions!

i) Hydroxyl ions in RE³⁺-doped SLN



- Above a threshold Er concentration a new OH⁻ band appears at ≈ 3488 cm⁻¹
- New OH⁻ bands appear for other RE³⁺ ions as well

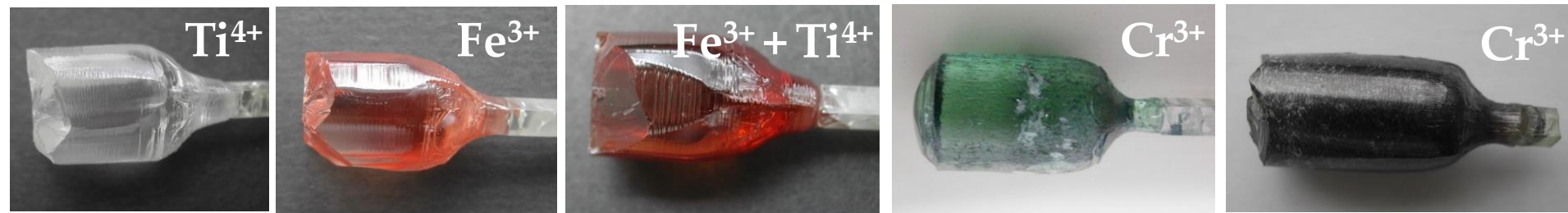
i) RE^{3+} ions in stoichiometric LiNbO_3



- RE^{3+} ions fit into the incorporation model of ODR ions

i) TM^{n+} ions in stoichiometric $LiNbO_3$

Ti^{4+} , Fe^{3+} , $Fe^{3+}+Ti^{4+}$ and Cr^{3+} doped SLN crystals were grown by the HTTSSG and Czochralski methods



SLN – HTTSSG
0.12 mol% Ti

SLN – HTTSSG
0.12 mol% Fe

SLN – HTTSSG
0.24 mol% Fe + 0.12 mol% Ti

SLN – HTTSSG
0.5 mol% Cr

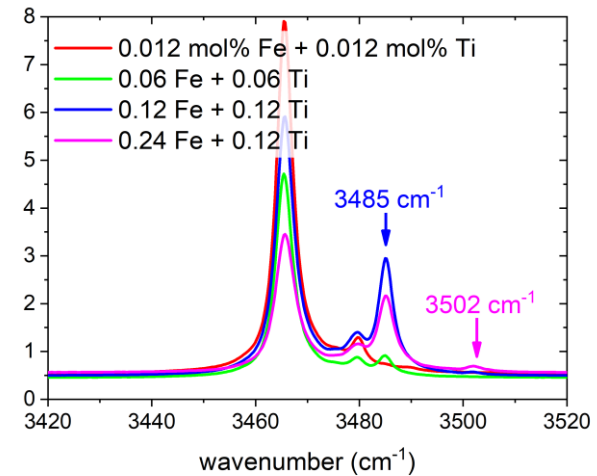
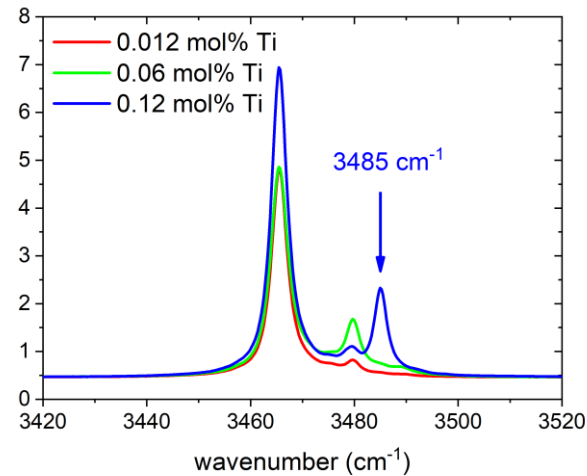
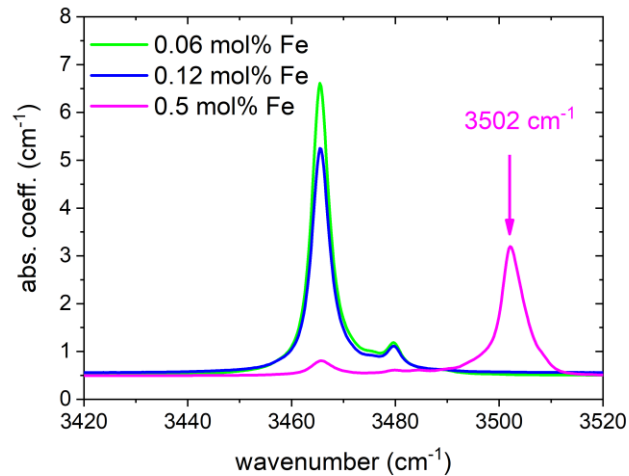
SLN – Czochralski
0.5 mol% Cr
Li/Nb=1.38

Sample series	Conc. in solution [mol%]		Conc. in crystal [mol%]		OH ⁻ band [cm ⁻¹]		Sample series	Growth method	Cr conc. in solution/melt [mol%]	Cr-OH band [cm ⁻¹]
	Fe	Ti	Fe	Ti	Fe	Ti				
1	0.06	0	-	-	-	-	4	HTTSSG	0.1	-
	0.12	0	0.056	-	-	-		HTTSSG	0.5	-
	0.5	0	0.23	-	3502	-		Czochralski	0.5	3502
2	0	0.012	-	-	-	-				
	0	0.06	-	-	-	-				
	0	0.12	-	0.077	-	3485				
3	0.012	0.012	-	-	-	-				
	0.06	0.06	-	-	-	3485w				
	0.12	0.12	0.068	0.069	3502vw	3485				
	0.24	0.12	0.13	0.048	3502w	3485				

L. Kovács, L. Kocsor, É. Tichy-Rács, K. Lengyel, L. Bencs, and G. Corradi: Hydroxyl ion probing transition metal dopants occupying Nb sites in stoichiometric $LiNbO_3$, *Optical Materials Express*, **9**, 4506-4516, 2019.

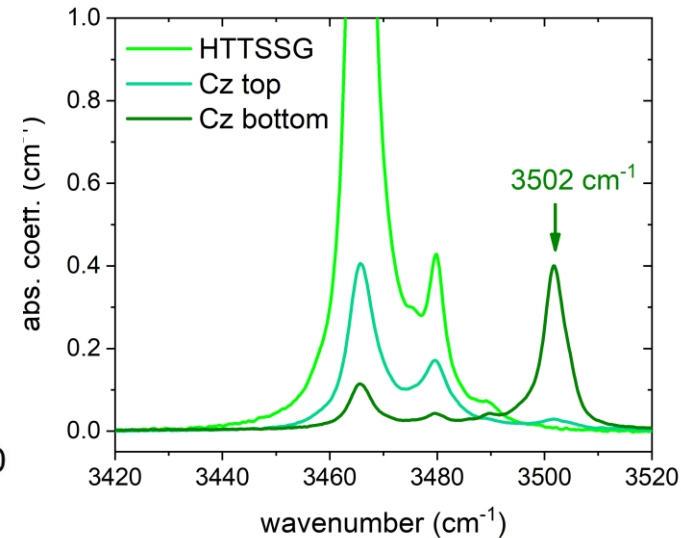
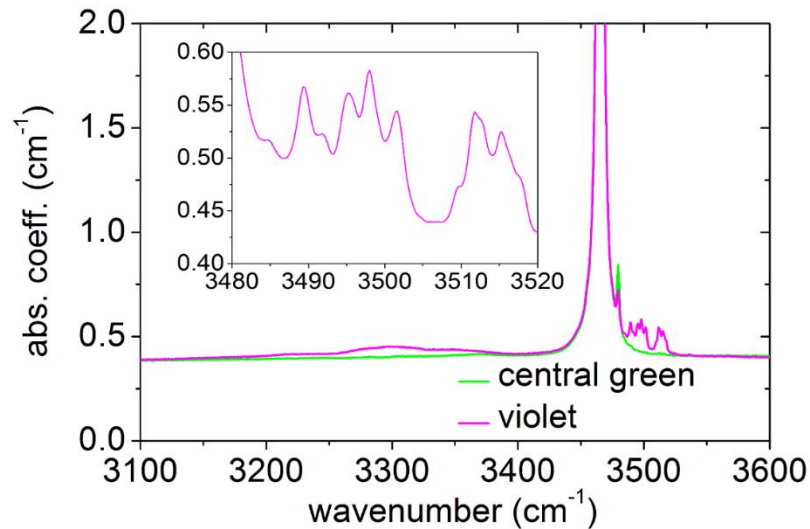
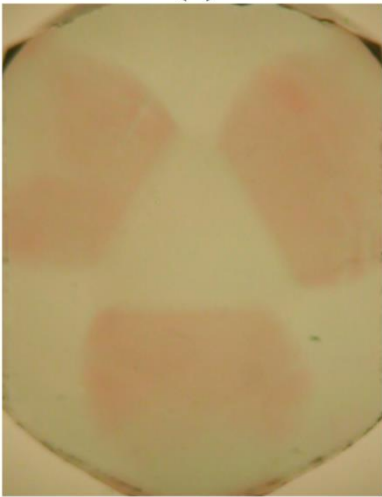
Crystals
2020

i) OH⁻ ions in Fe³⁺- and Ti⁴⁺-doped SLN



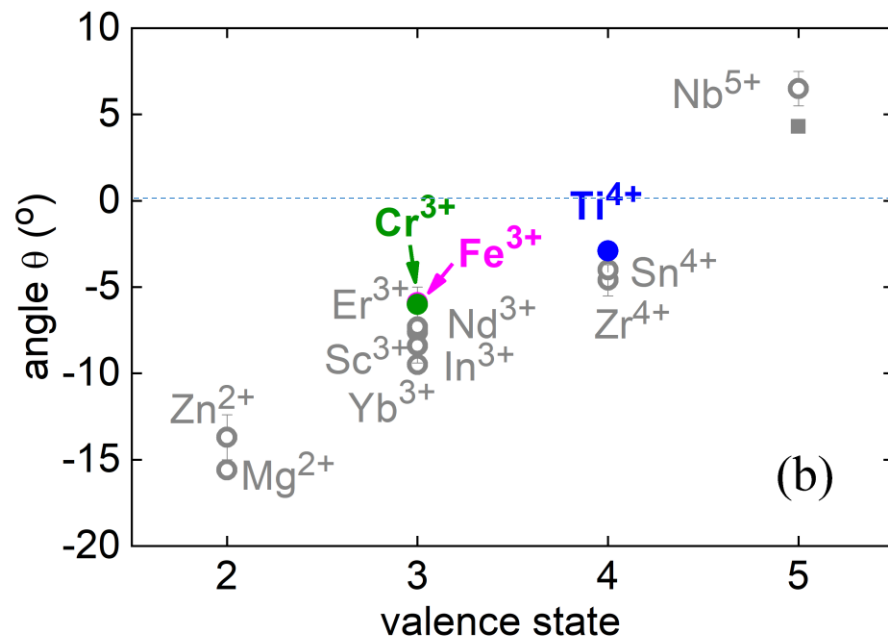
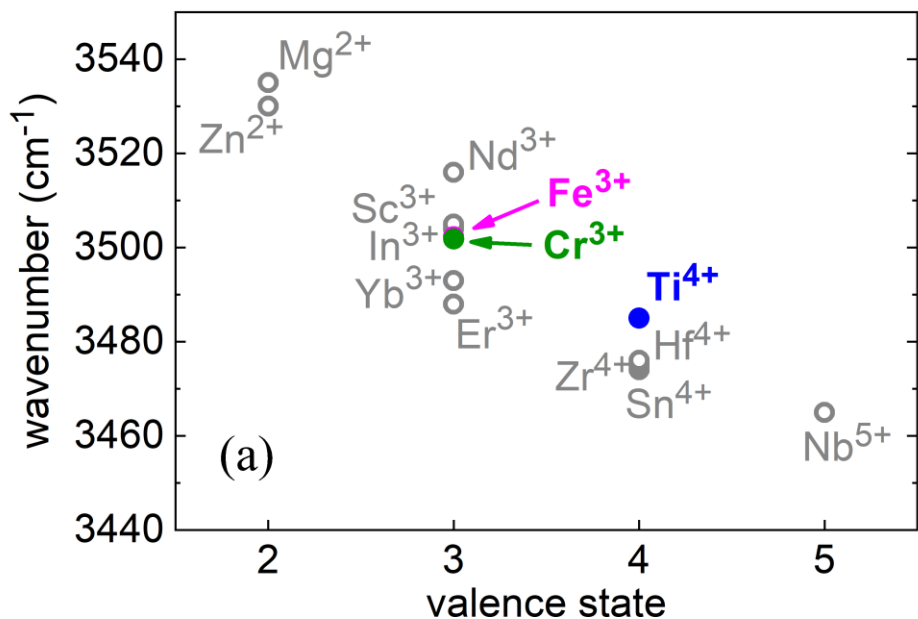
- Above a threshold concentration of the TMⁿ⁺ dopants **new OH⁻ absorption bands** appear in SLN crystals

i) Hydroxyl ions in Cr³⁺-doped SLN



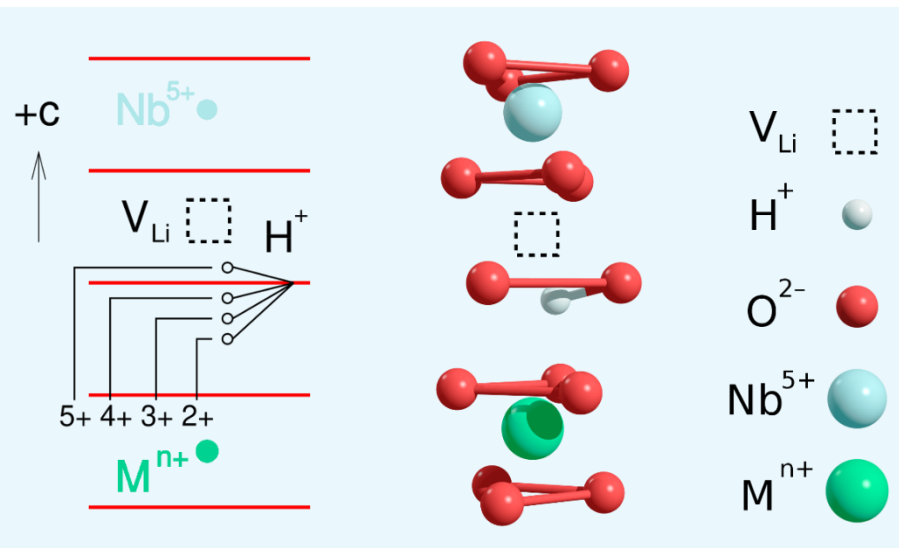
- In the inhomogeneous bottom part of Cr-doped SLN a number of narrow overlapping OH⁻ bands appeared due to Cr – OH centres with different defect environments
- Due to the composition change (becoming more stoichiometric) a single OH⁻ band appears at 3502 cm⁻¹ at the bottom part of the Czochralski-grown crystal similarly to the Fe-doped SLN

i) TM^{n+} ions in stoichiometric LiNbO_3



- The higher the valence state of the dopant, the lower the OH^- vibrational frequency and the closer the O–H bond direction to the oxygen plane
- **The observed trend is valid for all dopants studied so far**

i) M^{n+} ions in LiNbO_3 – Summary



Periodic Table of the Elements

The periodic table shows the following ions highlighted in colored circles:

- Mg^{2+} (green circle)
- Ti^{4+} (blue circle)
- Cr^{3+} (blue circle)
- Fe^{2+} (blue circle)
- Zn^{2+} (green circle)
- In^{3+} (blue circle)
- Sn^{2+} (red circle)
- Er^{3+} (blue circle)
- Yb^{2+} (blue circle)

- The IR absorption spectra of the OH^- stretching vibration have been investigated in stoichiometric LiNbO_3 doped with M^{n+} (ODR^{n+} , RE^{3+} , TM^{n+}) ions.
- Above a threshold M^{n+} dopant concentration new OH^- absorption bands appear due to the presence of $M^{n+}-\text{OH}^-$ type defects in the crystals. In the $M^{n+}-\text{OH}^-$ complex the dopant substitutes Nb site.
- It has been reaffirmed that OH^- ions are excellent probes of the defect structure in LiNbO_3 crystals.

ii) Mechanochemical reactions of LiNbO_3 induced by high-energy ball-milling

- Lithium niobate nanocrystals
- High-energy ball-milling (dry and wet grinding)
- Particle and grain size reduction
- Phase transformation and chemical reaction
- Structure characterization (X-ray, Raman, reflection spectroscopy, coulometric titration, electron microscopy)

ii) LiNbO₃ nanocrystals

- Can be used e.g. in
 - Nonlinear optics – as harmonic nanoparticles (HNP) in nano-biophotonics
 - Quantum optics – rare-earth doped LN as single photon source
- Can be prepared by
 - „Bottom up” method
 - Mechanochemical calcination (grinding + heat treatment)
 - Wet chemical, sol-gel, hydrothermal, combustion, etc.
 - „Top down” method
 - **High-energy ball-milling (dry and wet grinding) for particle and grain size reduction**
 - Shaker mill – Spex mixer mill
 - Planetary mill – Fritsch Pulverisette
 - Etching

ii) High-energy ball-milling

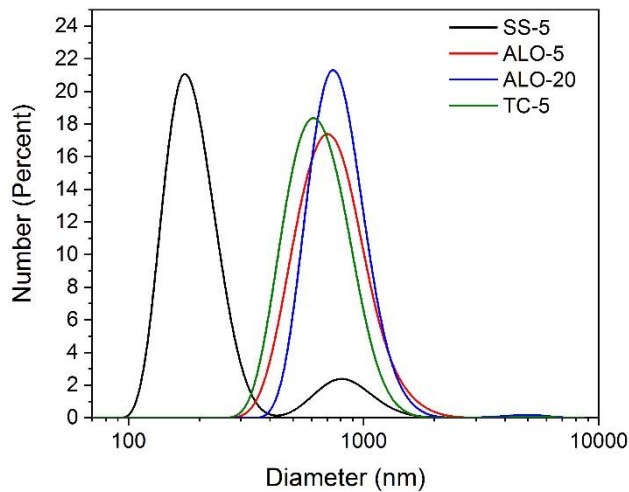
Dry grinding in SPEX shaker mill

Grinding parameters							
#	Vial	Ball	Time (h)	Number of balls	Ball-to-powder mass ratio	Ball-to-powder volume ratio	Sample quantity (g)
SS-5	Stainless steel	11 mm 5.5 g	5	2	3.8 : 1	2.2 : 1	2.9
ALO-5	Alumina	12.5 mm	5	2	3.8 : 1	4.4 : 1	2.2
ALO-20		4.2 g	20				
TC-5	Tungsten carbide	11 mm 10.7 g	5	2	3.8 : 1	1.2 : 1	5.65

L. Kocsor, L. Péter, G. Corradi, Z. Kis, J. Gubicza and L. Kovács: Mechanochemical reactions of lithium niobate induced by high energy ball-milling, *Crystals*, **9**, 334/1-14, 2019.

ii) High-energy ball-milling

Dry grinding in SPEX shaker mill



Particle diameter distributions of the ground samples in different vials determined by the Dynamic Light Scattering (DLS) method

#	Resulting particle diameter (nm)	Resulting grain diameter (nm)
	DLS	XRD
SS-5	190, (800)	55±18
ALO-5	700	63±21
ALO-20	700	37±2
TC-5	500	51±6

Particle and grain sizes of samples ground in different vials

ii) High-energy ball-milling

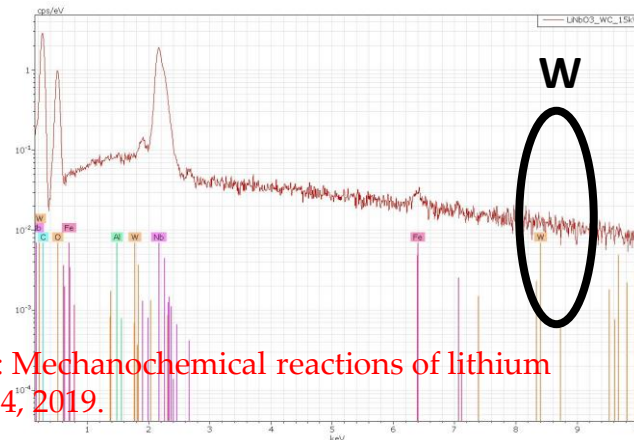
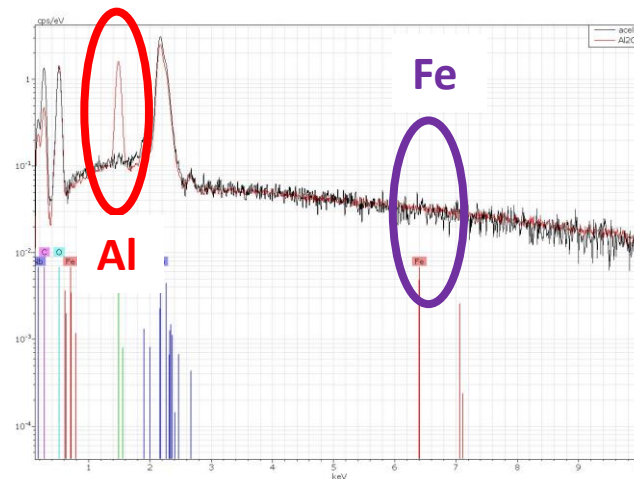
Dry grinding in SPEX shaker mill



Sample coloration

- **Al** was detected in the sample ground in **alumina** vial
- No **Fe** and **W** were detected in samples ground in **stainless steel** and **tungsten carbide** vials, respectively
- **The coloration is not related to contamination**

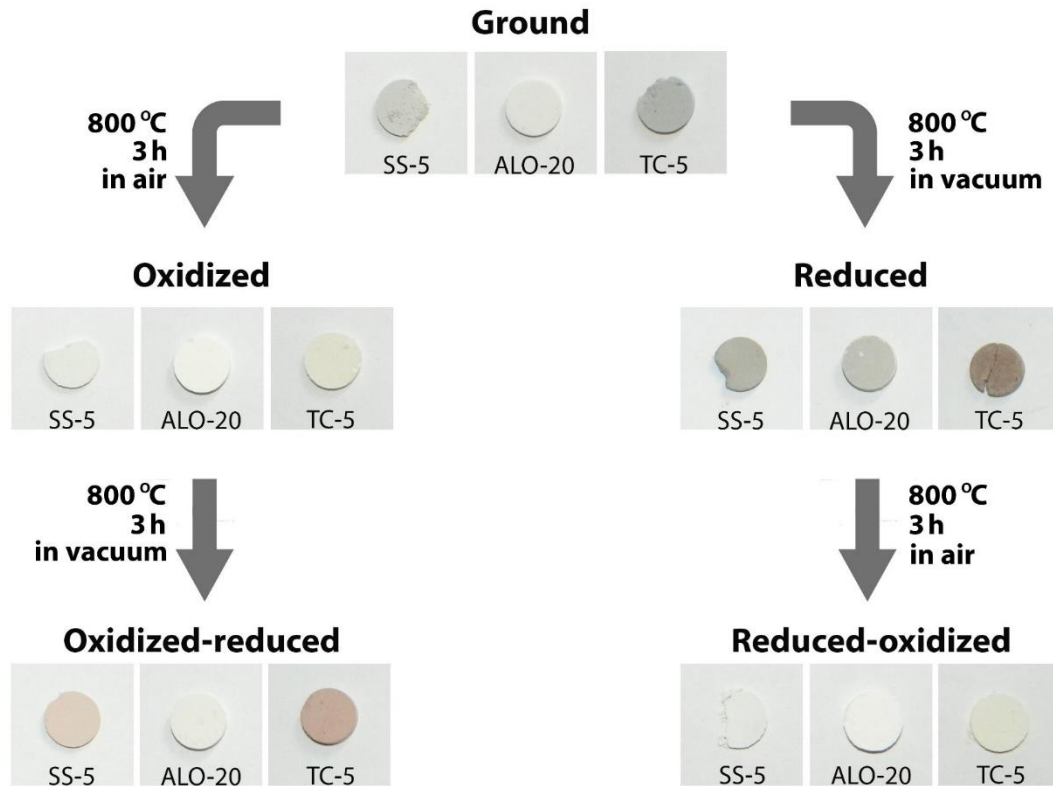
Energy-dispersive X-ray spectroscopy



L. Kocsor, L. Péter, G. Corradi, Z. Kis, J. Gubicza and L. Kovács: Mechanochemical reactions of lithium niobate induced by high energy ball-milling, *Crystals*, 9, 334/1-14, 2019.

ii) High-energy ball-milling

Dry grinding in SPEX shaker mill



The series of samples ground in different vials with subsequent heat treatments

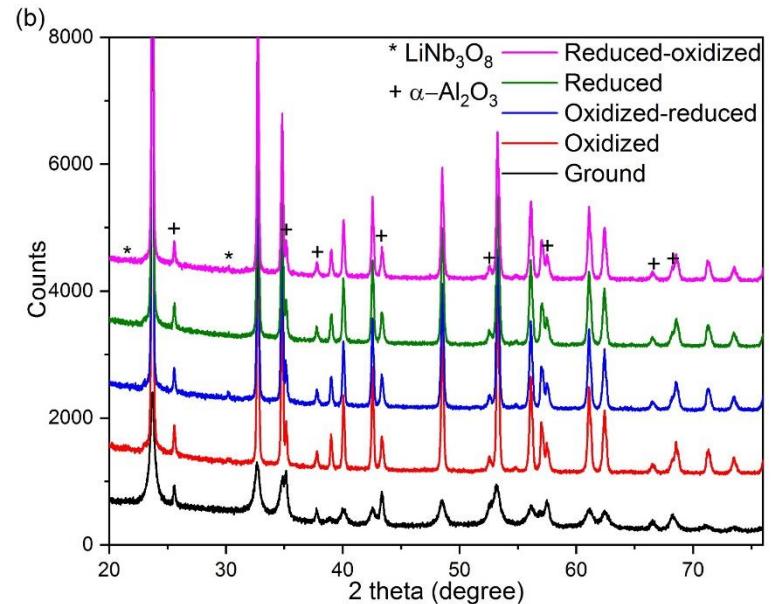
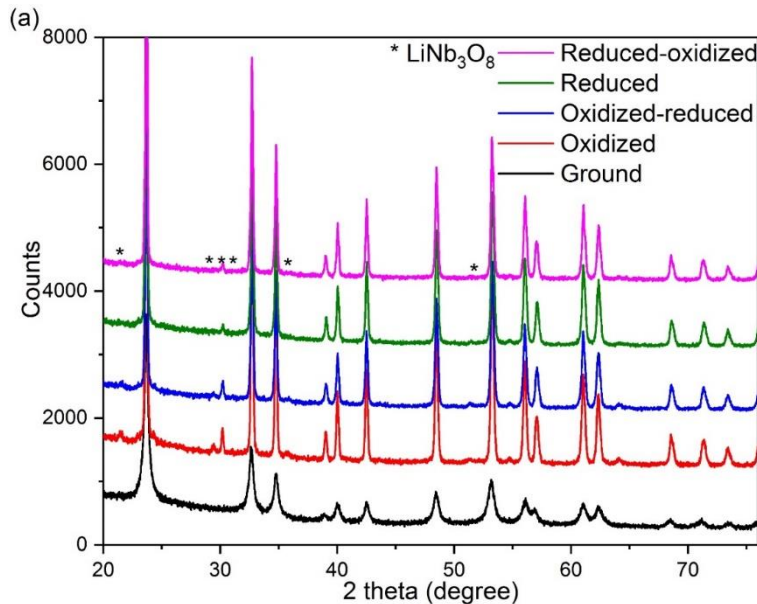
The samples underwent a change concerning the oxidation state of niobium during the grinding process.

Annealing treatments in oxidative or non-oxidative atmospheres was applied for restoring or modifying the oxidation state of niobium in the ground samples.

ii) High-energy ball-milling

Dry grinding in SPEX shaker mill

X-ray diffraction



Diffraction patterns of LN ground in **stainless steel** (a) and **alumina** (b) vials

The unmarked peaks are the reflections of LiNbO_3

- As-ground samples – broad peaks – small grain size
- Heat-treatment results in narrower diffraction lines due to increased grain sizes
- LiNb_3O_8 phase appeared in annealed samples



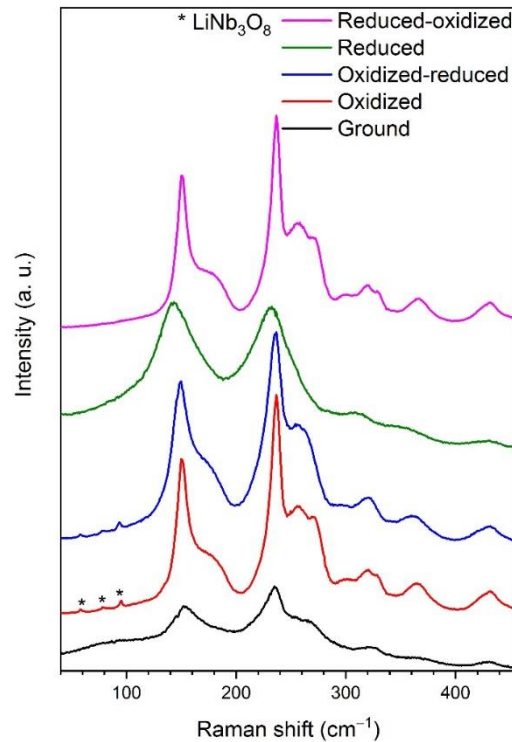
L. Kocsor, L. Péter, G. Corradi, Z. Kis, J. Gubicza and L. Kovács: Mechanochemical reactions of lithium niobate induced by high energy ball-milling, *Crystals*, **9**, 334/1-14, 2019.

ii) High-energy ball-milling

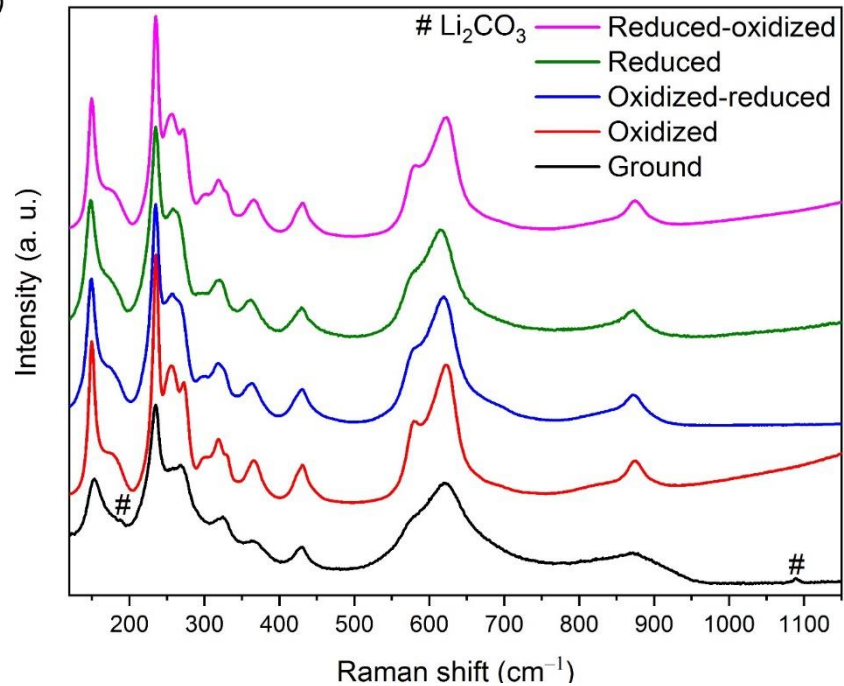
Dry grinding in SPEX shaker mill

Raman spectroscopy

(a)



(b)

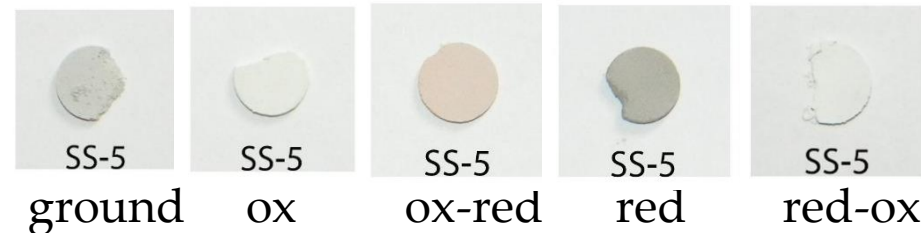
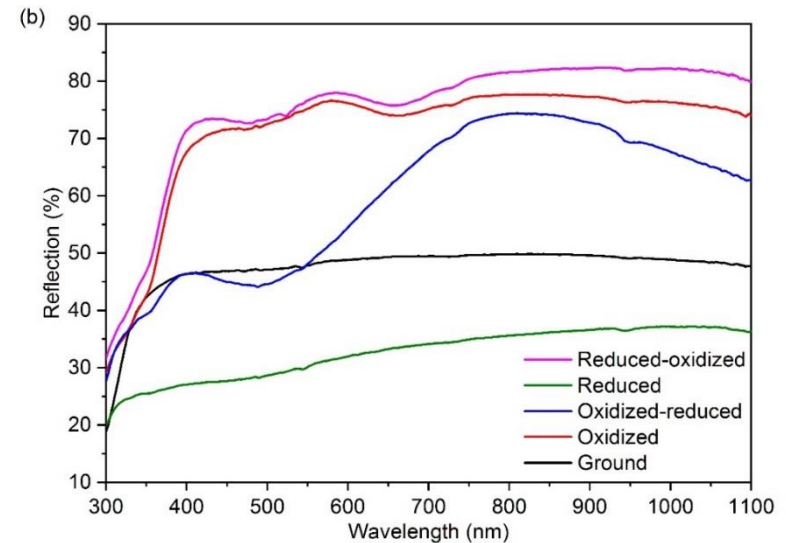
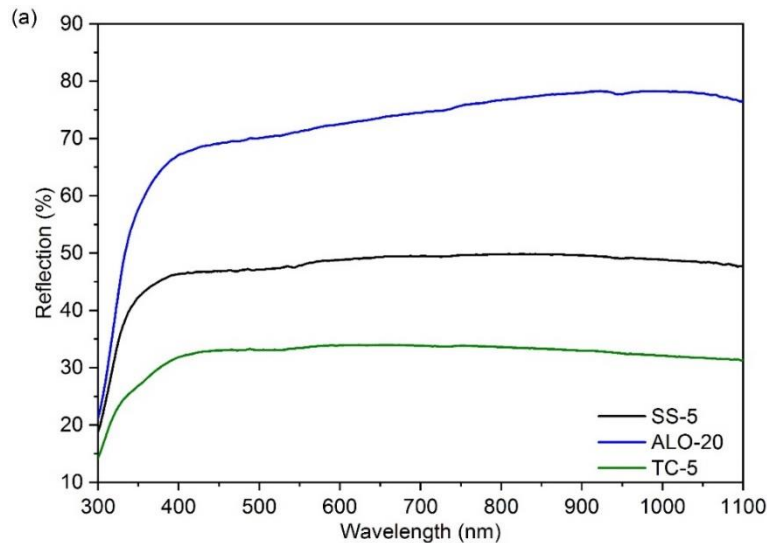


Raman spectra of ground and heat-treated samples ball-milled for 5 h in **stainless steel** (a), and in **tungsten carbide vial** (b)

L. Kocsor, L. Péter, G. Corradi, Z. Kis, J. Gubicza and L. Kovács: Mechanochemical reactions of lithium niobate induced by high energy ball-milling, *Crystals*, **9**, 334/1-14, 2019.

ii) High-energy ball-milling

Dry grinding in SPEX shaker mill
Optical reflection measurements



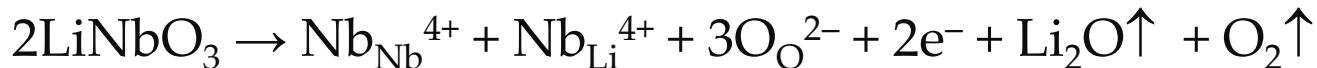
Optical reflection spectra of samples ball-milled in different vials (a) and of ground and heat-treated samples ball-milled in **stainless steel** vial (b)

ii) High-energy ball-milling

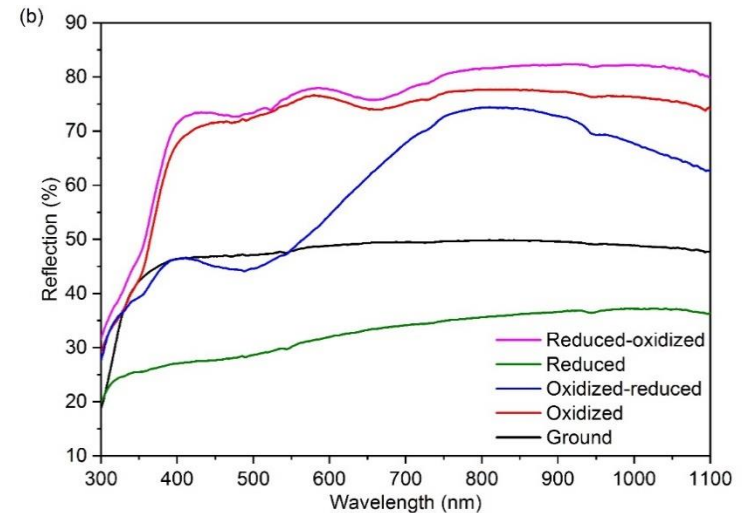
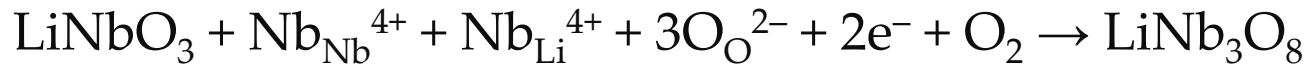
Dry grinding in SPEX shaker mill

- Congruent LiNbO_3 crystals:
 - $\text{Li}_{1-5x}\text{Nb}_{1+x}\text{O}_3$, $x \approx 0.01$
 - $\text{Nb}_{\text{Li}}^{4+}$: small polaron
 - $\text{Nb}_{\text{Li}}^{4+} - \text{Nb}_{\text{Nb}}^{4+}$: bipolaron

- Ball-milling



- Oxidation



ii) High-energy ball-milling

Dry grinding in SPEX shaker mill

Quantitative determination of the degree of decomposition during ball-milling

- Coulometric titration $m_{\text{OX}} = MQ/2F$
 - M : molar weight of Li_2O
 - Q : charge passed until the equivalence point
 - F : Faraday-constant (96485 C/mol)
 - 2: the hydrolysis of 1 mol of Li_2O results in 2 mols of hydroxide ions

Sample	100 w_{ox} (weight%)	Mean particle radius, DLS (nm)
ALO-20	0.97±0.05	350
TC-5	1.05±0.10	250
SS-5	1.52±0.21	95

The measured Li_2O mass ratios w_{ox} , expressed as weight percent of the as-ground powder

ii) High-energy ball-milling

Dry grinding in SPEX shaker mill

Estimation of the thickness of the LiNb_3O_8 layer

- Assumptions
 - Uniform spherical LN particles of unmodified composition
 - Uniformly thick LiNb_3O_8 layer
 - $d \ll R$

$$w_{\text{OX}} = \frac{m_{\text{LTN}} \frac{M_{\text{OX}}}{M_{\text{LTN}}}}{m_{\text{LN}} + m_{\text{LTN}} \left(1 + \frac{M_{\text{OX}}}{M_{\text{LTN}}}\right)}$$
 weight ratio of Li_2O in the ground material

$$d_{\text{LTN}} \approx \frac{R}{3} \frac{\rho_{\text{LN}}}{\rho_{\text{LTN}}} \frac{M_{\text{LTN}}}{M_{\text{OX}}} w_{\text{OX}} \quad \text{and} \quad d_{\text{OX}} \approx \frac{\rho_{\text{LTN}}}{\rho_{\text{OX}}} \frac{M_{\text{OX}}}{M_{\text{LTN}}} d_{\text{LTN}} \approx \frac{d_{\text{LTN}}}{5.6}$$

Sample	$100w_{\text{ox}}$ (weight%)	Mean particle radius, DLS (nm)	Mean grain radius, XRD (nm)	d_{LTN} LiNb_3O_8 shell thickness (nm)	d_{ox} Li_2O shell thickness (nm)
ALO-20	0.97 ± 0.05	350	18.5	14.6	2.6
TC-5	1.05 ± 0.10	250	25.5	11.3	2.0
SS-5	1.52 ± 0.21	95	27.5	6.2	1.1

L. Kocsor, L. Péter, G. Corradi, Z. Kis, J. Gubicza and L. Kovács: Mechanochemical reactions of lithium niobate induced by high energy ball-milling, *Crystals*, **9**, 334/1-14, 2019.

ii) High-energy ball-milling

Dry and wet grinding in Fritsch planetary mill

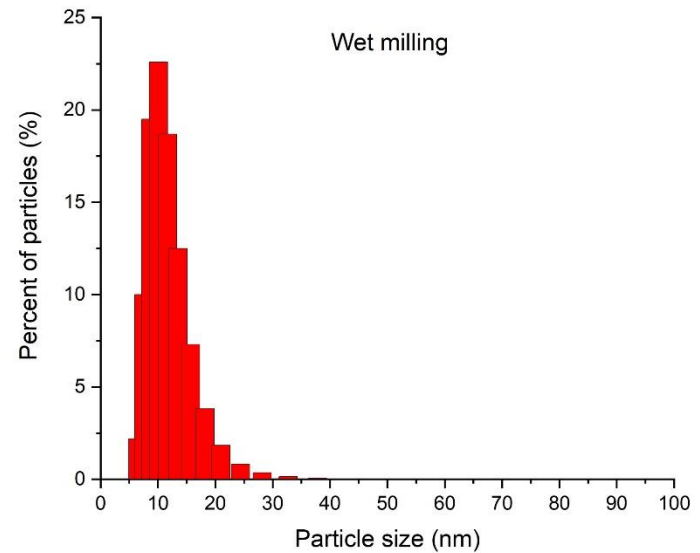
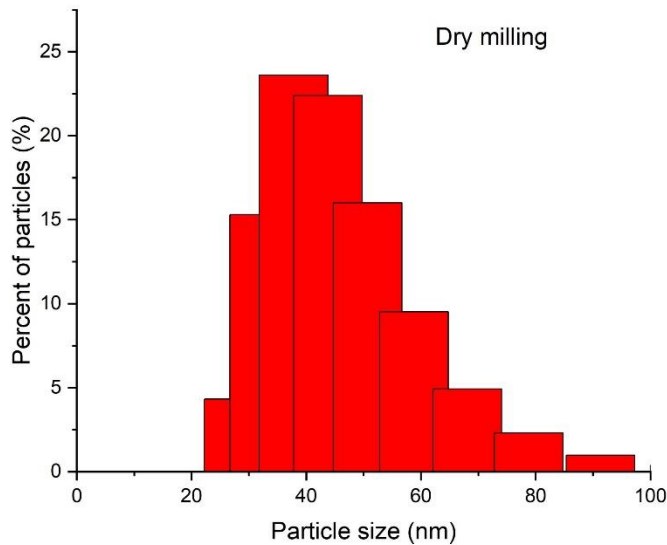
Recent results

Grinding parameters						
	Vial / Ball	RPM	Time	Solvent	Ball size / quantity	Sample quantity
DRY	ZrO ₂	1100	10x1 min	-	0.5-3 mm / 70 g	5 g
WET	ZrO ₂	1100	10x1 min	10 ml water	0.1-3 mm / 70 g	5 g

ii) High-energy ball-milling

Dry and wet grinding in Fritsch planetary mill

Recent results

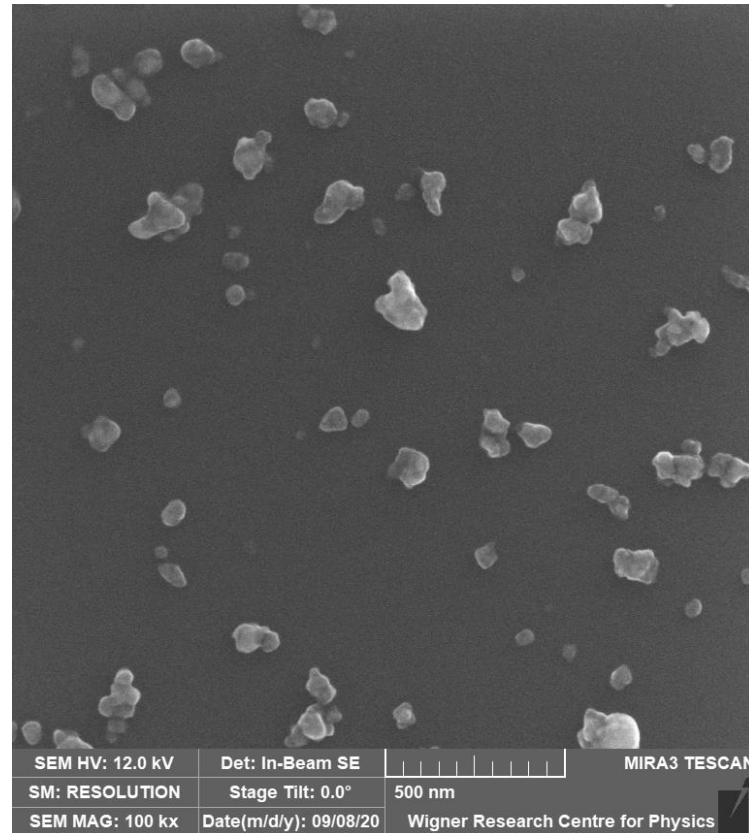
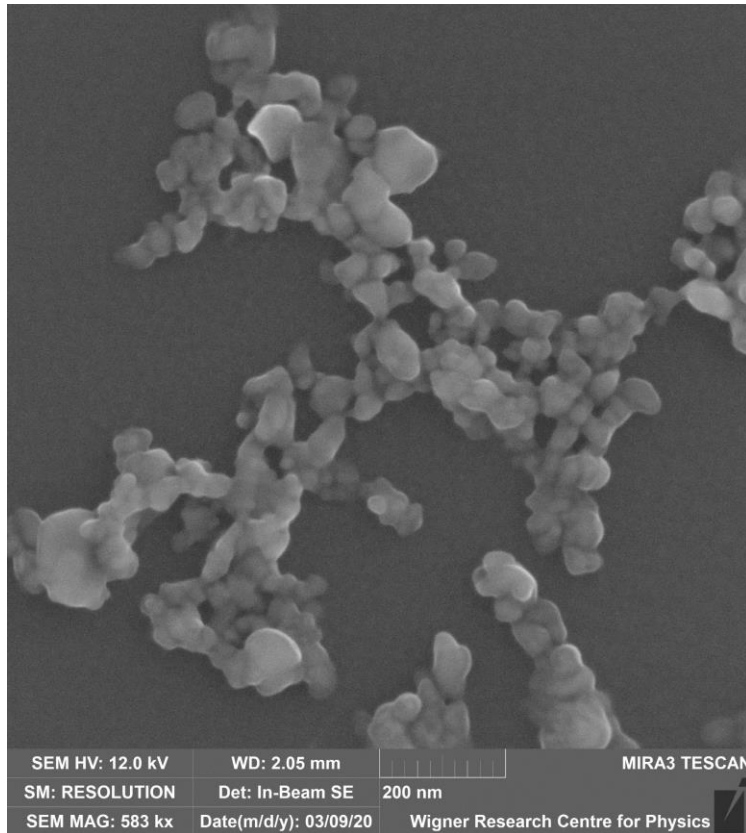


Size distribution measured by Dynamic Light Scattering (DLS)

ii) High-energy ball-milling

Dry grinding in Fritsch planetary mill

Recent results

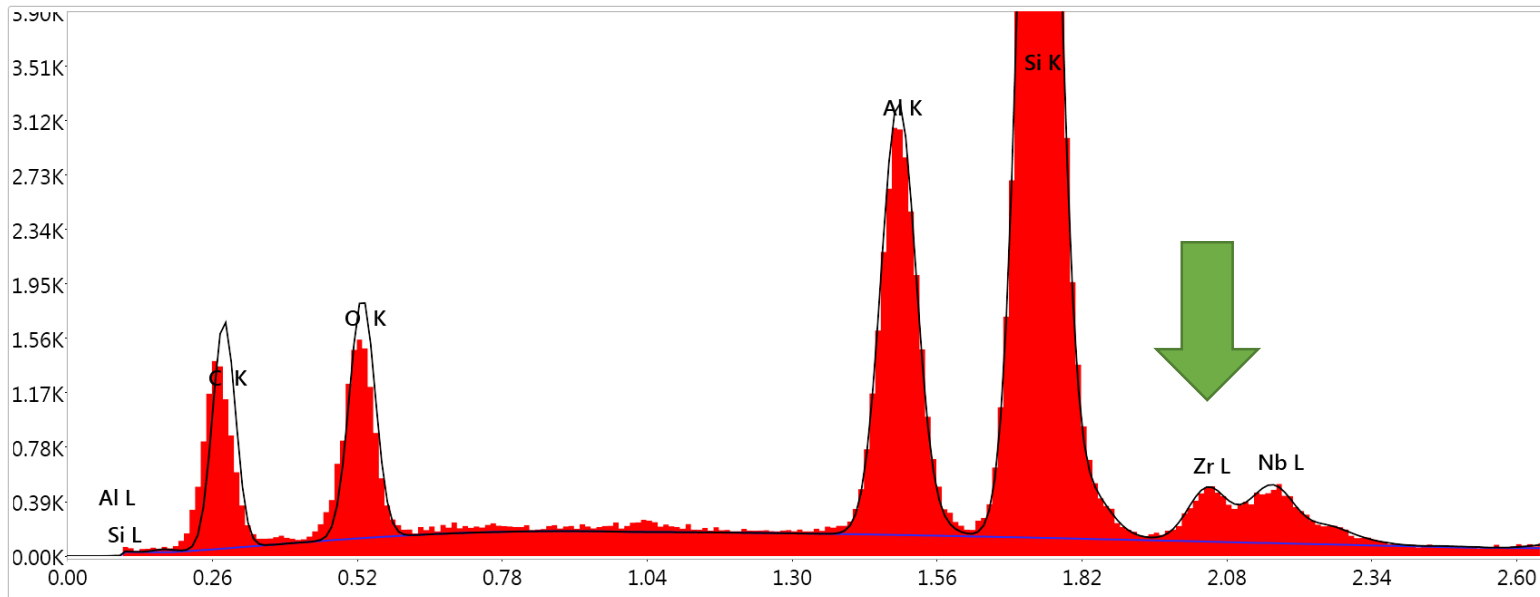


Scanning electron microscopy (SEM) images of the LiNbO₃ particles

ii) High-energy ball-milling

Wet grinding in Fritsch planetary mill

Recent results



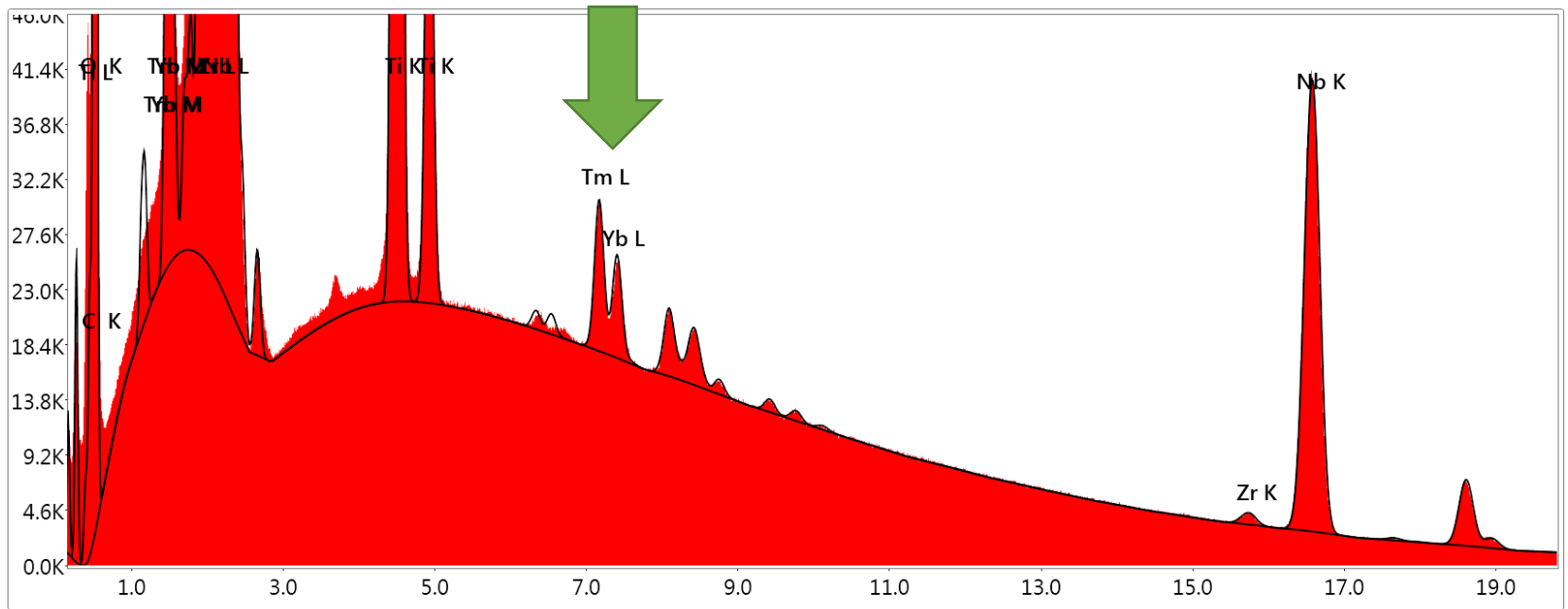
Det: Element-C2B

The material (Zr) of the vial/ball is present in the particles measured by EDAX

ii) High-energy ball-milling

Dry grinding in Fritsch planetary mill

Recent results



Date: 11/01/2011

Dopants (Tm and Yb) planned to use in single photon source are present in the particles

ii) High-energy ball-milling – Summary

- Nano-LN (10-50 nm particle size) has been successfully prepared by high-energy ball-milling
- During the milling process the material suffers partial reduction interpreted by polaron/bipolaron formation.
- Li_2O loss and LiNb_3O_8 formation was observed in the outer shell of the particles
- These findings provide a comprehensive explanation of the physicochemical behaviour of the system during grinding and annealings.
- **RE^{3+} -doped nano-LN has been successfully prepared for single photon source**

Conclusions

- **LiNbO₃ either in single crystal or in nanocrystalline form is one of the most versatile materials for optical applications**
- **The study of both intrinsic and extrinsic defects affecting the properties of LN may provide new results for future applications**

Acknowledgment



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