

$\text{Li}_{1+y}\text{Ti}_{2-x-y}\text{Ge}_x\text{Al}_y(\text{PO}_4)_3$ NASICON-type electrolytes with enhanced conductivity

for solid state lithium-ion batteries

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The use of NASICON-type phosphates in lithium-ion batteries

advantages:

- ✓ More resistant to high temperatures,
- ✓ Less flammable
- ✓ More electrochemically stable

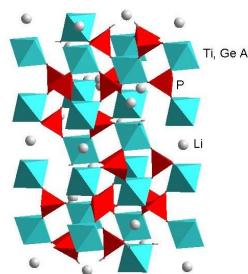
disadvantages:

- ✓ Low ionic conductivity
- ✓ Poor contact of the electrode and electrolyte

Lithium-titanium phosphates - $\text{LiTi}_2(\text{PO}_4)_3$
Space group R-3c; $a = 8.5129 \text{ \AA}$, $c = 20.8780 \text{ \AA}$
Ionic conductivity $\sigma < 10^{-6} \text{ S/cm at } 50^\circ\text{C}$

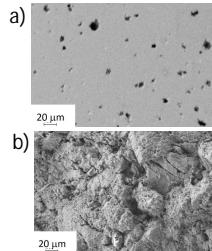
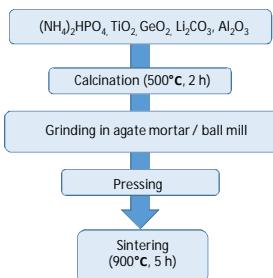
Enhancing ionic conductivity:
Iso- and heterovalent doping

The aim of this research:
Synthesis and investigation of ionic conductivity
 $\text{Li}_{1+y}\text{Ti}_{2-x-y}\text{Ge}_x\text{Al}_y(\text{PO}_4)_3$ ($y = 0-0.3$)



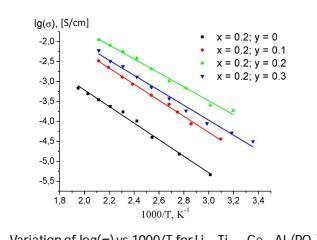
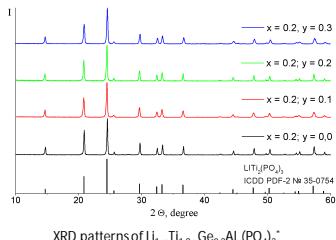
Crystal structure of $\text{Li}_{1+y}\text{Ti}_{1.6}\text{Ge}_{0.2}\text{Al}_{0.2}(\text{PO}_4)_3$

Synthesis scheme



SEM-images of polished pellets of $\text{Li}_{1+y}\text{Ti}_{1.6}\text{Ge}_{0.2}\text{Al}_{0.2}(\text{PO}_4)_3$ prepared with different modes of precursor ball-milling:
a) 400 rpm, 99 cycles; b) 600 rpm, 99 cycles

Influence of heterovalent doping of $\text{LiTi}_2(\text{PO}_4)_3$ on ionic conductivity



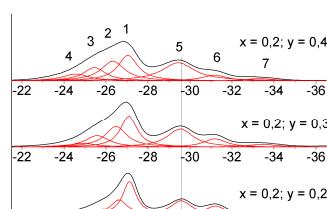
Unit cell parameters of $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$ (space group R-3c)

y	a, Å	c, Å
0	8.4924(5)	20.851(4)
0.1	8.4774(4)	20.876(2)
0.2	8.4752(8)	20.880(5)
0.3	8.4594(4)	20.833(2)

y	E_a , kJ/mol
0	40 ± 1
0.1	38 ± 1
0.2	31 ± 1
0.3	36 ± 1

* The highest ionic conductivity of $\text{LiTi}_2(\text{PO}_4)_3$ was found for $x = 0.2$ / Kurzina E. A., Stenina I. A., Dalvi A., Yaroslavtsev A.B. Synthesis and Ionic Conductivity of Lithium Titanium Phosphate-Based Solid Electrolytes. *Inorg. Mater.* 57 (2021) 1035–1042

^{31}P MAS NMR of $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$

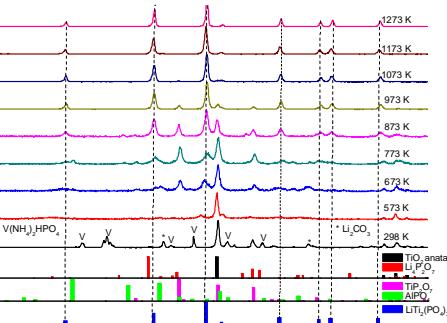


Chemical shifts of the 5th peak:

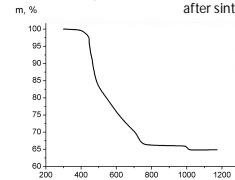
X = 0.2; y = 0.2 $\delta = -29.57 \text{ ppm}$
 X = 0.2; y = 0.3 $\delta = -29.53 \text{ ppm}$
 X = 0.2; y = 0.4 $\delta = -29.45 \text{ ppm}$

Environment	Peak area, %			
	X = 0,2	Y = 0,2	Y = 0,3	Y = 0,4
1 PO_4Ti_4	31,04	24,54	22,59	
2 $\text{PO}_4\text{Ti}_3\text{Al}$	16,92	22,36	20,63	
3 $\text{PO}_4\text{Ti}_2\text{Al}_2$	6,93	9,08	13,07	
4 PO_4TiAl_3	8,96	8,87	8,12	
5 $\text{PO}_4\text{Ti}_2\text{Ge} + \text{PO}_4\text{Ti}_2\text{GeAl}$	19,50	21,43	26,08	
6 $\text{PO}_4\text{Ti}_2\text{Ge}_2$	11,46	8,00	6,40	
7 PO_4TGe_3	4,12	4,07	2,91	

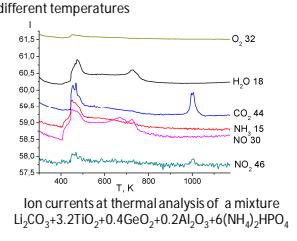
Investigation of sintering process of $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$



XRD patterns of a mixture $\text{Li}_2\text{CO}_3+3.2\text{TiO}_2+0.4\text{GeO}_2+0.2\text{Al}_2\text{O}_3+6(\text{NH}_4)_2\text{HPO}_4$ at 298 K and after sintering for 5 h at different temperatures

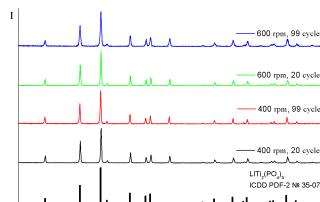


Weight loss curve of a mixture $\text{Li}_2\text{CO}_3+3.2\text{TiO}_2+0.4\text{GeO}_2+0.2\text{Al}_2\text{O}_3+6(\text{NH}_4)_2\text{HPO}_4$

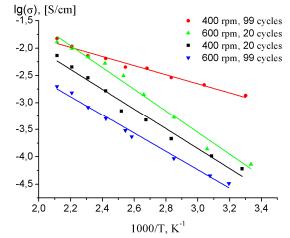


Ion currents at thermal analysis of a mixture $\text{Li}_2\text{CO}_3+3.2\text{TiO}_2+0.4\text{GeO}_2+0.2\text{Al}_2\text{O}_3+6(\text{NH}_4)_2\text{HPO}_4$

Influence of precursor ball-milling on ionic conductivity of $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$



XRD patterns of $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$ prepared with different modes of precursor ball-milling



Plots of $\log(s)$ vs $1000/T$ for $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$ prepared with different modes of precursor ball-milling

Rate of ball milling, rpm	Number of cycles	Pellet density, %	Conductivity (298 K), S/cm	E_a , kJ/mol
400	20	85	$6.6 \cdot 10^{-5}$	35 ± 2
400	99	97	$1.4 \cdot 10^{-3}$	16 ± 2
600	20	88	$7.3 \cdot 10^{-5}$	38 ± 2
600	99	73	$3.3 \cdot 10^{-5}$	32 ± 1

Conclusions:

- ✓ $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$ compounds were prepared by solid state reaction and characterized by X-ray diffraction, scanning electron microscopy, ^{31}P MAS NMR and impedance spectroscopy.
- ✓ The optimal germanium content has been determined. The aluminum doping results in increase in lithium conductivity.
- ✓ $\text{Li}_{1.2}\text{Ti}_{1.6}\text{Ge}_{0.2}\text{Al}_{0.2}(\text{PO}_4)_3$ exhibits the highest ionic conductivity at room temperature ($1.4 \cdot 10^{-4} \text{ S/cm}$).
- ✓ Detected ^{31}P NMR signals have been assigned to different $\text{PO}_4\text{Ti}_{4-n-m}\text{Ge}_n\text{Al}_m$ arrangements assuming a random distribution of Ti, Al, and Ge cations in NASICON compounds. The intensity of the components located at less negative values becomes more intense as the Al content increases.
- ✓ The formation of $\text{Li}_{1+y}\text{Ti}_{1.8-y}\text{Ge}_{0.2}\text{Al}_y(\text{PO}_4)_3$ occurs through the intermediate formation of titanium pyrophosphate.
- ✓ The precursor ball-milling before the final annealing leads to a conductivity increase.