

Proceeding

# Opportunities for Reversible Bonding at the Cellulose/Matrix Interface of Composites by Mussel-inspired Interface Design<sup>†</sup>

Pieter Samyn <sup>1,\*</sup>

<sup>1</sup> SIRRIS – Department Innovations in Circular Economy, Gaston Geenslaan 8, B-3001 Leuven, Belgium

\* Correspondence: pieter.samyn@outlook.be.

<sup>†</sup> Presented at the 4<sup>th</sup> International Electronic Conference on Applied Sciences, 22 October–10 November 2023;

Available online: [https:// https://asec2023.sciforum.net/](https://https://asec2023.sciforum.net/)

**Abstract:** With increasing demand for recycling composite materials and re-use of fibers and matrix as new resources in the context of circular economy, composite materials often pose problems as they form complex aggregates. The user properties of high strength and long lifetime require strong interfaces between the matrix and reinforcing fibers, while recycling would benefit from easy separation of both phases. Therefore, the design of an interface with reversible bonding upon thermal or chemical activation may offer a good balance. In addition, the request for bio-based composites incorporating cellulose fibers should be combined with bio-inspired interface modification avoiding traditional chemical surface modification. An impressive example of reversible bonding in nature is observed by mussels and regulated by so-called mussel-foot proteins. The latter include dopamine as a main component that presents reversible bonding upon change in pH. In present work, cellulose fibers were modified with a dopamine (DA) or polydopamine (PDA) coating that was polymerized in contact with the cellulose surface, providing good chemical compatibility and interaction with cellulose hydroxyl groups. The adhesive properties of modified cellulose fibers were investigated by local adhesive measurements through atomic force microscopy and vary between strong adhesion (low pH) and weak adhesion (high pH). In parallel, the macroscale mechanical strength of epoxy composites with modified fibers improved while the interface adhesion of the modified fibers dropped after submersion in solutions with pH = 9.8. Based on these observations, a proof of concept for recycling of cellulose/epoxy composites and recovery of cellulose fibers is demonstrated after grinding and chemical treatment at high pH.

**Keywords:** composite; interface; debonding; mussel-inspired

**Citation:** To be added by editorial staff during production.

Academic Editor: Firstname Last-name

Published: date



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## 1. Introduction

Composites offer excellent combinations of mechanical properties and lightweight through the crucial selection of their components, including a polymer matrix and reinforcing fibers. Their performance particularly relies on a strong interface between fiber and matrix through maximizing the chemical interaction after surface modification [1], e.g., through silanization or alternative methods. Driven by scarcity in resources, in parallel, the circular economy requires more proficient use of materials through incorporation of renewables and recycling them into pure fractions. The end-of-life scenarios for cellulose-based composites are often complex as they contain multiple materials that are explicitly well integrated with each other to provide best performance, while incorporation of long cellulose fibers as a bio-based reinforcing phase provides strength and stiffness [2]. In a circular economy, however, it remains a crucial challenge how to disassemble and recycle high-performance cellulose-based composites [3].

The mechanisms for reversible bonding with a transition between strong and weak adhesion strength depending on external triggers are exploited in adhesives and bondable joints of fiber-reinforced plastics [4]: possible debonding technologies include

thermally expanding microspheres, induction heating, foaming agents or chemical degradation in combination with a variety of reversible chemistries. An impressive example of reversible bonding is observed in nature, regulated through the mussel-foot proteins (*Mefp*) [5], which are enriched in adhesive catechol groups of dopamine. Various synthetic polymers have been functionalized with a catechol offering diverse adhesive, sealant, coating, or anchoring properties [6]. The reversible molecular interactions and binding mechanisms of *Mefp* were elucidated through force measurements and translated into the design of self-healing composites [7]. The general principles of mussel-inspired surface modification as a functionalization method for polymers were further exploited to introduce reinforcement, flame retardance [8], or conductivity [9]. Alternatively, modification of NiTi and Ti-V-Al wires with adhesive proteins yield better macroscale properties of composites through promotion of the interfacial adhesion [10]. The ecofriendly surface modification of bamboo fiber through self-assembly of a polydopamine layer showed better compatibility with the polybutylene succinate (PBS) matrix, opening a wide range of applications for natural fiber-reinforced composites [11]. Moreover, the good compatibility and interactions between polydopamine and polysaccharide materials [12], or modification of cellulose with polydopamine in particular [13], was extensively reviewed before. Therefore, new pathways should be explored for the design of functional materials and interfaces by combining these biomaterials. However, most attention at present was put on interface compatibilization and bonding of matrix and fiber, while introducing debonding effects with dopamine were not yet firmly investigated.

Below, a feasibility study is presented to illustrate the possibilities for reversible adhesive bonding between cellulose fibers and an epoxy resin matrix, which is promising to provide both high interface strength and debonding properties after dissolving the composite material at suitable pH. The better recovery of modified cellulose fibers from the matrix may provide a novel method to enhance recyclability of composite materials.

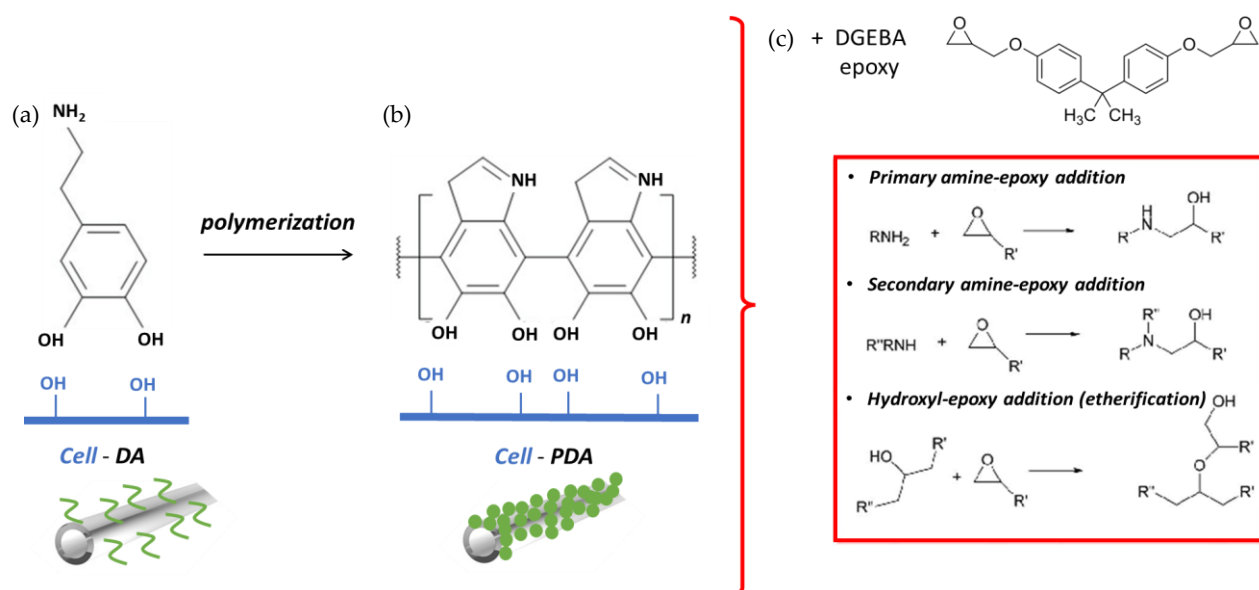
## 2. Materials and Methods

### 2.1. Cellulose materials and modification

The cellulose fibers originated from Kraft hardwood pulp with cellulose fiber diameters of 20 to 25  $\mu\text{m}$ . The pulp fibers were modified through deposition of a coating layer with dopamine (Cell-DA) or polydopamine (Cell-PDA), according to Figure 1. The dopamine hydrochloride powder was purchased from Merck and dissolved in water in a concentration of 1 mg/ml, in presence of a buffer solution at neutral pH (Cell-DA) or alkaline pH (Cell-PDA). A 2 wt.-% suspension of the cellulose pulp was added into the solution and mixed overnight under continuous stirring on a magnetic plate at room temperature. The slurry was subsequently washed with water and filtered for recovery of the modified fibers, which were further dried in a hot-air oven at 60°C for 30 minutes.

The samples of Cell-DA were obtained after mixing at neutral pH = 7.2 conditions while maintaining the non-polymerized dopamine structure. The latter allows for the assembly of the hydroxyl groups towards the cellulose surface through hydrogen bonding and accessibility of the primary amine groups (Figure 1a). The interaction mechanisms between dopamine and cellulose were reviewed before [13], and include absorption and physical interaction through hydrogen bonding.

The assembled dopamine layer was converted into a polymerized coating on the cellulose surface of Cell-PDA samples, introduced through an oxidative self-polymerization of the dopamine under known weak alkaline conditions at pH = 8.5 in 10 mM tris(hydroxymethyl)-aminomethane (TRIS) buffer. During reaction, a polymerized layer forms, composed of individual PDA particles deposited onto the cellulose surface (Figure 1b). The success of the polymerization reaction was visually observed through the black color of the solution.



**Figure 1.** Cellulose fiber modification and formulation of composites with epoxy matrix, (a) physical layer deposit with dopamine layer (Cell-DA), (b) coating with polydopamine (Cell-PDA), (c) mixing of modified cellulose fibers in epoxy matrix.

## 2.2. Composite formulation

A composite was formulated by mixing 20 wt.-% and 30 wt.-% Cell, Cell-DA or Cell-PDA fibers (w/w) in an epoxy matrix and casting the blend as a film. Sample codes are as: epoxy+Cell20, epoxy+Cell30, epoxy+Cell-DA20, epoxy+Cell-DA30, epoxy+Cell-PDA20, epoxy+Cell-PDA30. The epoxy resin was formulated by mixing bisphenol A diglycidyl ether or DGEBA (EP101, Resion Resin Technology, Moordrecht, The Netherlands) with an amine hardener of 3-aminomethyl-3,5,5-trimethylcyclohexylamine (30 to 50 wt.-%) and m-phenylene bis(methylamine) (10 to 30 wt.-% (EP113, Resion Resin Technology, Moordrecht, The Netherlands)). The resin and hardener were mixed in 1:1 stoichiometric ratio as calculated from the epoxy equivalent weight EEW = 200 g/eq and amine hydrogen equivalent weight AHEW = 100 g/eq, followed by the incorporation of the cellulose fibers during mixing for 5 minutes.

The composite formulations were cast as thin films (2 mm) in an aluminum dish and thermally cured overnight at 80°C in a hot circulating air oven. During curing, the presence of primary amine, secondary amine and hydroxyl groups aid in the crosslinking reaction of the epoxy resin (Figure 1c), with highest reactivity of a primary amine addition.

## 2.3. Characterization methods

The intrinsic adhesive properties of DA under different pH conditions towards a cellulose fiber were probed through AFM adhesion measurements with a chemically modified AFM tip ( $\text{Si}_3\text{N}_4$ ). The tip was functionalized by grafting of DA and submerged in buffer solutions of pH = 4 and pH = 9.5. The tip was lowered at constant rate until it makes contact under a maximum load of 30 nN with a native cellulose fiber that was fixed on a glass support. The force-displacement curve was recorded during approach and retraction of the cantilever, allowing to calculate the adhesive force from the characteristic point of separation between the tip and the surface. The measurements were repeated 20 times in order to draw a histogram and determine intrinsic adhesive forces.

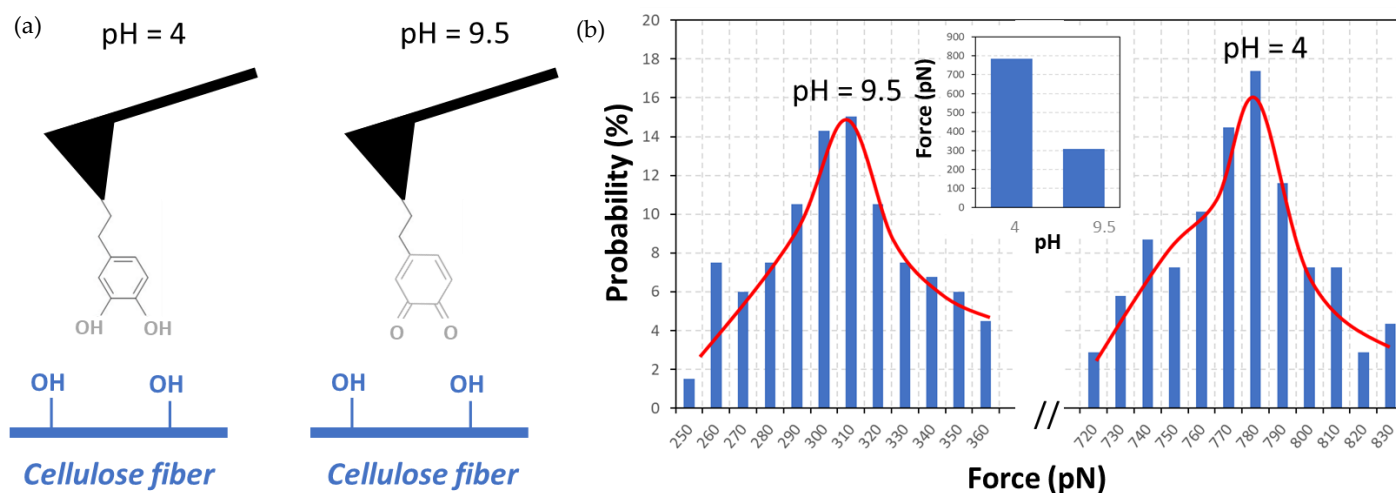
The tensile strength of the cellulose/epoxy composites was determined from standard mechanical testing of composite films (ASTM D3039). Microscopic images were taken on a TM3000 tabletop scanning electron microscope (Hitachi, Krefeld, Germany).

The fiber recovery was determined after mechanical grinding and dissolution of the epoxy films in an aqueous buffer solution at pH = 4 and pH = 9.5, followed by centrifugation and weight determination of the recovered fiber fraction after oven drying.

### 3. Results and Discussion

#### 3.1. Reversible adhesive properties towards cellulose fibers

The fundamentals for reversible bonding at the fiber/matrix interface of composites rely on the intrinsic interactions between both components under different chemical environments. The interactions between a functionalized AFM tip and cellulose fibers were probed in aqueous buffer solutions at pH = 4 and pH = 9.5 to identify the effects of the functional groups in dopamine under hydrolyzed conditions (dopa) or oxidized conditions (dopa-quinone), according to the set-up in Figure 2a. Similar experiments for determination of single-molecule adhesion of mussel-inspired adhesives towards metal or TiO<sub>2</sub> surfaces were previously made [14], indicating reversible bonding of dopamine depending on pH. Indeed, also for the adhesive contact with a cellulose fiber surface, two force ranges were detected with weak adhesion ( $300 \pm 20$  pN) at high pH and strong adhesion ( $775 \pm 40$  pN) at low pH (Figure 2b). Whereas metal complexation plays a role in contact with inorganic substrates, the interactions through hydrogen bonding with cellulose fiber surfaces at neutral to low pH are efficiently demonstrated. The switch in adhesion of dopamine towards cellulose at high pH raises opportunities for reversible bonding [5].

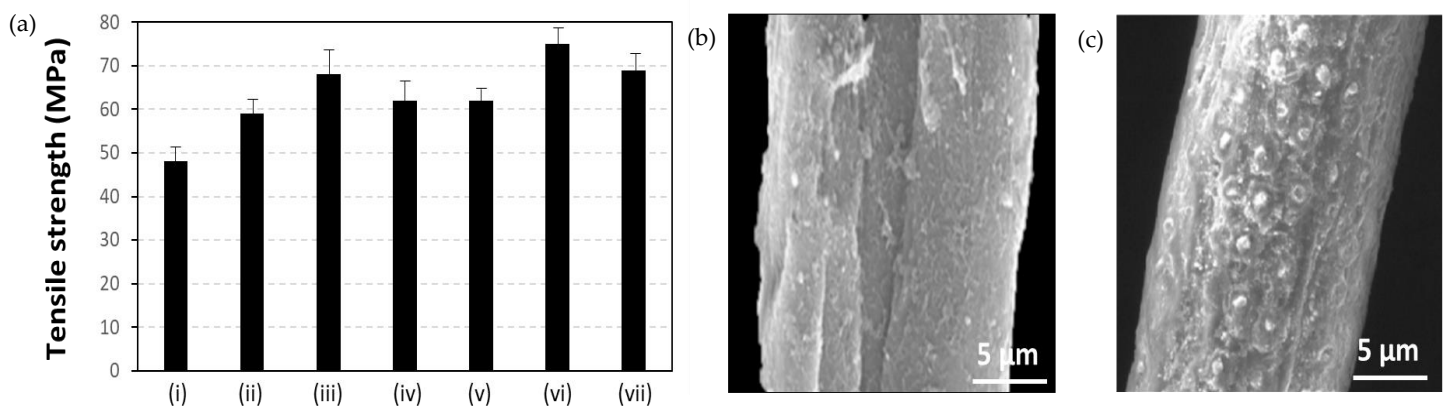


**Figure 2.** Adhesive measurements of DA molecules towards cellulose fibers under different pH conditions through approach and retraction experiment with modified AFM tip, (a) test set-up with positioning of modified AFM tip and native cellulose fiber, (b) distribution of adhesive forces measured during retraction at pH = 4 and pH = 9.5.

#### 3.2. Composite properties

The effects of surface modification of cellulose fibers on the mechanical properties of the epoxy composites were evaluated by measuring maximum tensile strength at break (Figure 3a) for different composite compositions. The interface modification clearly improved strength properties of the composite relatively to unmodified fibers, with a higher mechanical strength for epoxy composites reinforced with Cell-DA fibers relatively to Cell-PDA fibers, which is confirmed for both fiber concentrations of 20 wt.-% and 30 wt.-%. The high strength of the epoxy+Cell-DA can be attributed either to the efficient hydrogen bonding at the interface, or good accessibility of the free primary amine that aids in crosslinking with the epoxy resin. The surface features of modified fibers are visualized

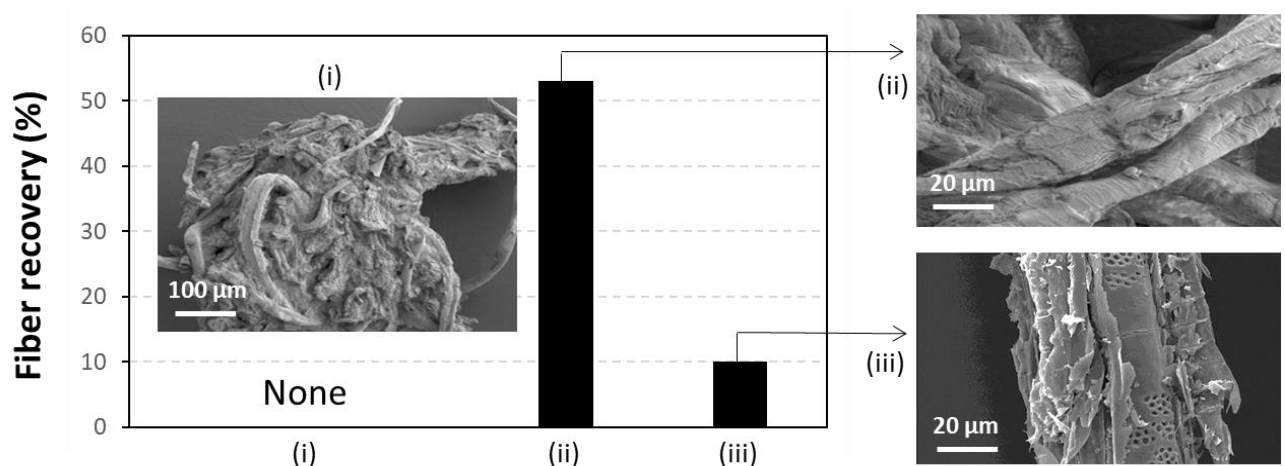
as thread-like deposits for Cell-DA (Figure 3b), or arranged particles for Cell-PDA (Figure 3c) in parallel with known morphologies for polydopamine coatings [12].



**Figure 3.** Evaluation of composite properties with modified cellulose fibers, (a) maximum tensile strength, for (i) epoxy matrix, (ii) epoxy+Cell20 (iii) epoxy+Cell-DA20 (iv) epoxy+Cell-PDA20 (v) epoxy+Cell30 (vi) epoxy+Cell-DA30 (vii) epoxy+Cell-PDA30, (b) SEM picture of modified cellulose fiber Cell-DA20, (c) SEM picture of modified cellulose fiber Cell-PDA20.

### 3.3. Composite debonding

The recovery of cellulose fibers after grinding and chemical dissolution at pH = 9.5 is quantified by weight measurement of recovered fibers, together with a qualitative image of recovered fibers from epoxy composites with 20 wt.-% Cell, Cell-DA or Cell-PDA (Figure 4). After dissolution, the unmodified fibers remain clogged within the epoxy matrix, while the Cell-PDA fibers were only partially recovered with indeed large fractions of the epoxy matrix remaining sticking at the cellulose surface. Better fiber recovery was noticed for modified fibers Cell-DA, with > 50 % recovery yield and smooth surface topography of the recovered fibers. The DA layer provides reversible interaction near the cellulose surface and allows for detachment of the epoxy. Experiments with dissolution at pH = 4 did not yield favorable recovery with fibers equally clogged in the epoxy.



**Figure 4.** Experimental observation for fiber recovery and respective morphologies after mechanical grinding and chemical dissolving epoxy/cellulose composites in buffer solution at pH = 9.5, for (i) epoxy+Cell20, (ii) epoxy+Cell-DA20, (iii) epoxy+Cell-PDA20.

## 4. Conclusion

This feasibility study demonstrated a route for the design of composite interfaces with reversible bonding between cellulose fibers and epoxy resin, both improving interface strength and debonding. The idea for bio-based adhesive surface modification of fibers through mussel-inspired functional groups was transferred into a synthetic material.

The intrinsic adhesive bonding between catechol groups and cellulose fibers was pH dependent, with strong adhesion at low pH and weak adhesion at high pH, in parallel with the oxidation of the catechol groups into quinone. The nanoscale adhesion could be employed for the surface modification of cellulose fibers and tuning the macroscale interaction with an epoxy resin, by comparing the interface characteristics after surface modification of the fibers with a dopamine (DA) or polydopamine (PDA) coating. The interaction with DA maximized the strength in the composite material while simultaneously allowing for better recovery of fibers after grinding and solvent treatment at high pH.

The results are promising to enhance recyclability of composite materials, where the separation between fiber and matrix constituents is often a hurdle when aligning composite design within a circular economy. Only one case study for thermoset materials has been presented here, which in future research should be further broadened towards thermoplastic composites.

**Author Contributions:** Conceptualization, experiments and data analysis, writing-original draft and revision P.S.; project administration P.S.; funding P.S.

**Funding:** This research was funded through the Robert Bosch Foundation (2012-2016) at University of Freiburg, Sustainable Use of Natural Resources, project Foresnab.

**Institutional Review Board Statement:** Not applicable.

**Informed Consent Statement:** Not applicable.

**Data Availability Statement:** Not applicable.

**Conflicts of Interest:** The author declares no conflict of interest.

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