

Evaluation of the corrosion behavior of low-temperature nitrided AISI 316L austenitic stainless steel

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INTRODUCTION & AIM

When **nitriding** is carried out at **low temperature** (<450 °C) on austenitic stainless steels, nitrogen atoms can freely diffuse in the metal lattice, while the diffusion of substitutional atoms, such as chromium, is significantly slowed down. As a consequence, **the formation of chromium nitride is hindered**, and hence the corrosion resistance can be preserved. Nitrogen atoms are retained in the fcc lattice of austenite well beyond their solubility limit, forming a **supersaturated solid solution**, known as **expanded austenite** (γ_N) or **S-phase**. The corrosion behavior of the nitrided layers depends on their microstructure and phase composition, the environment characteristics, and even on the testing conditions.

The **aim** of the present study is to evaluate the **corrosion behavior** of nitrided AISI 316L in **NaCl solution**, and, in particular, to assess its **repassivation capability**.

MATERIALS & METHODS

Material: AISI 316L (40×17×0.7 mm, polished up to 6 μm diamond paste)

Nitriding treatment: dc glow-discharge process

T: 380 °C; p: 130 Pa; t: 5 h; 80 %_{vol} N₂ + 20 %_{vol} H₂

Microstructure characterization

- Light microscopy
- X-ray diffraction analysis (XRD) (Bragg-Brentano configuration, CuK α)
- Knoop surface microhardness (load: 25 g_f)

Corrosion testing

Solution: 5 wt.% NaCl, aerated

Surface area: 1 cm²

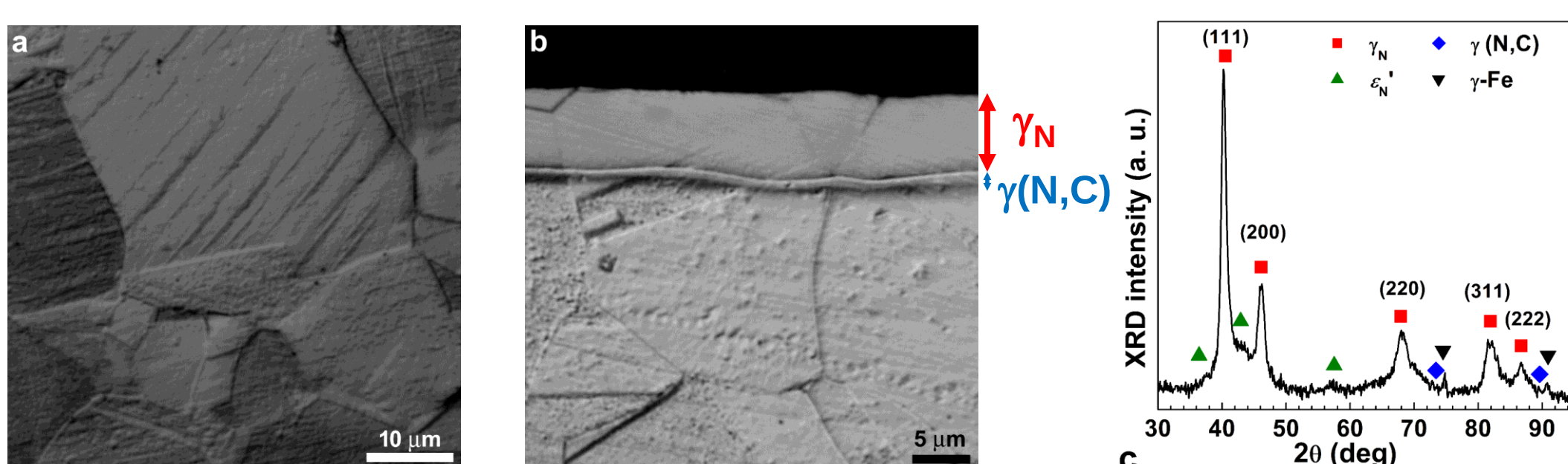
Delay before the test: 20 h

Test Methods:

- **Electrochemical Impedance Spectroscopy (EIS)** (at open circuit potential (OCP); ac amplitude (peak-to-peak): 2.5 mV; f: 10kHz-12mHz)
- **Cyclic potentiodynamic polarization** (rate: 0.3, 1 mV s⁻¹; polarization up to 100 $\mu\text{A cm}^{-2}$)
- **Galvanostatic test** (anodic current: 100 $\mu\text{A cm}^{-2}$; t: 3000 s)

RESULTS & DISCUSSION

Microstructure, Phase Composition & Microhardness

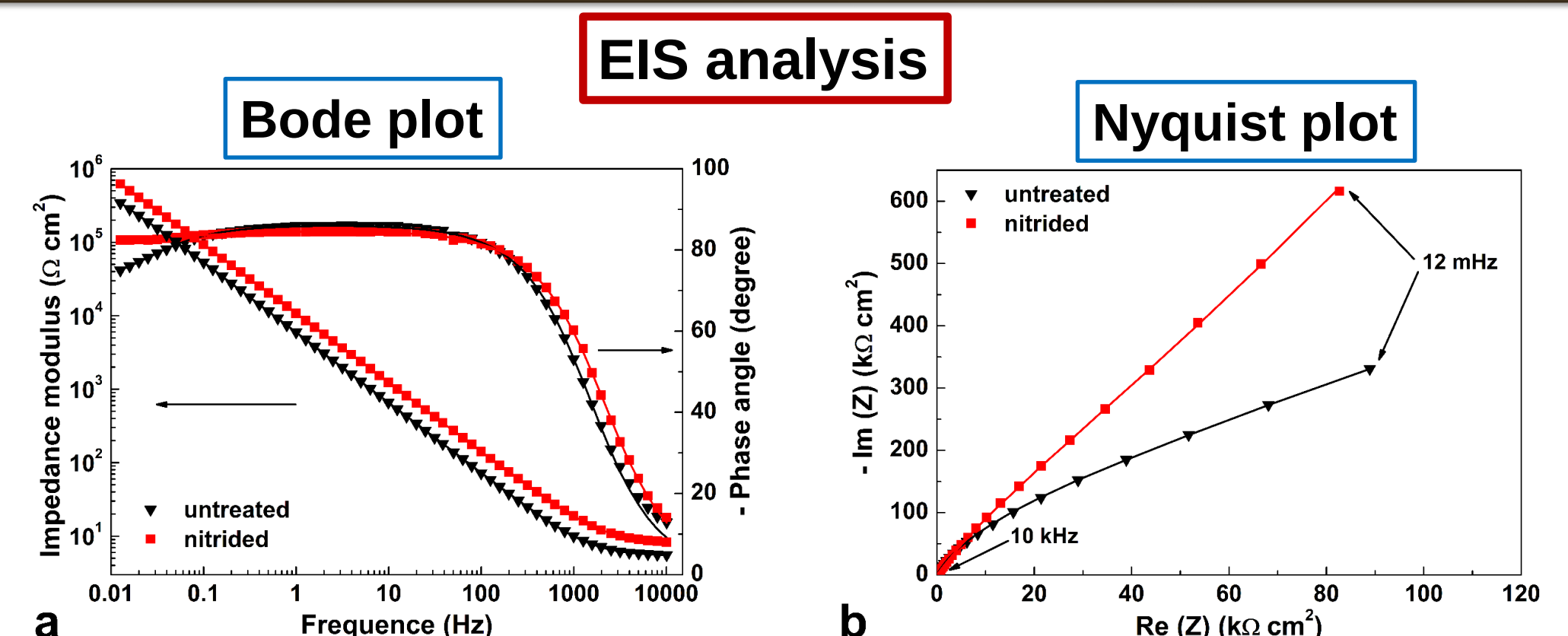


The formation of the nitrided layer causes **local plastic deformations** observable at the **surface** (a). The **nitrided layer** (b) consists of expanded austenite, γ_N , a less expanded austenite, $\gamma(\text{N,C})$, and a N-rich hcp martensite, ϵ'_N (c), due to a fcc-to-hcp martensitic transformation.

Thickness of the nitrided layer ($\gamma_N + \gamma(\text{N,C})$): $7.8 \pm 0.5 \mu\text{m}$

Surface hardness: $1473 \pm 108 \text{ kg}_f \text{ mm}^{-2}$ (untreated: $268 \pm 4 \text{ kg}_f \text{ mm}^{-2}$)

RESULTS & DISCUSSION



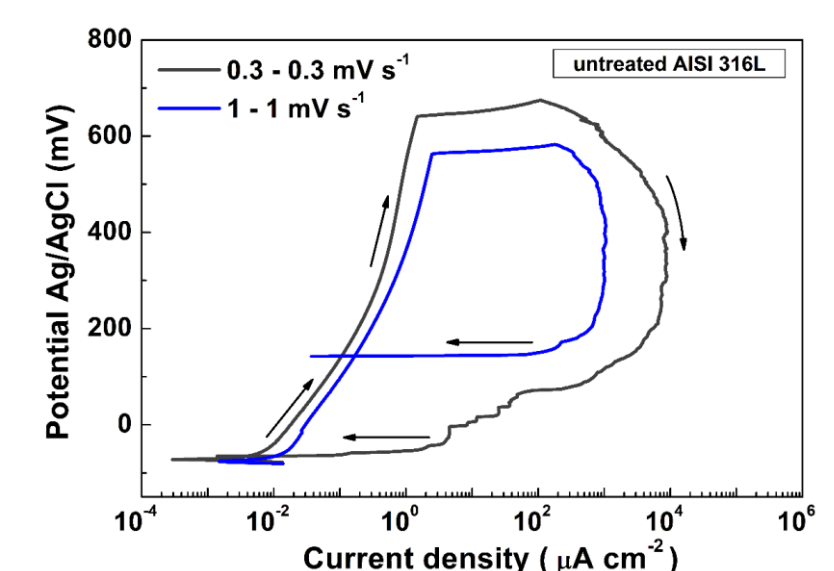
The **nitrided samples** have a **higher resistance to general corrosion**.

Cyclic potentiodynamic polarization

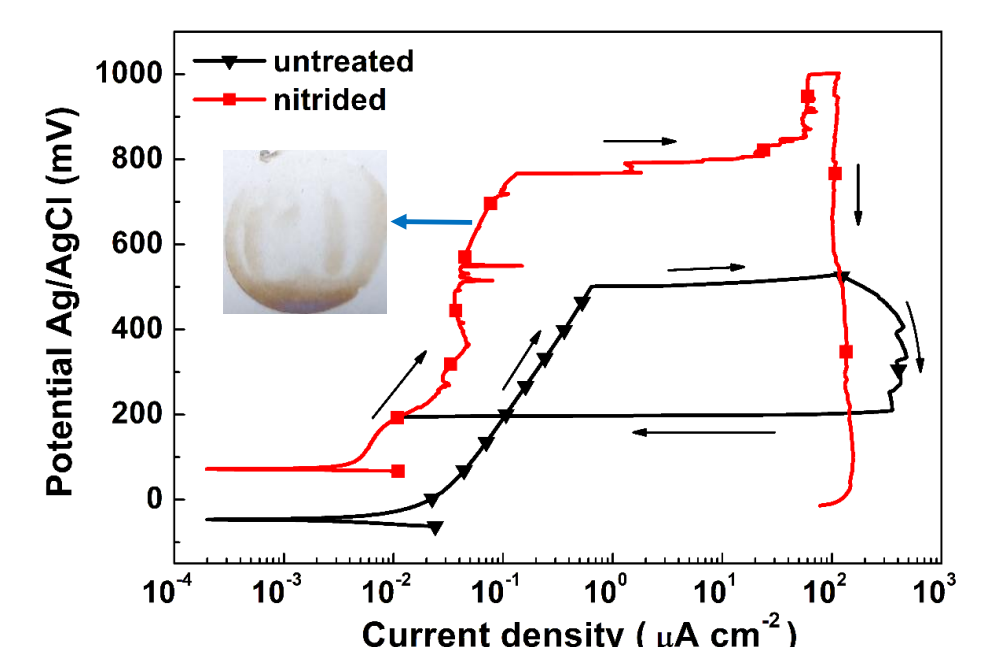
As observed for the untreated steel, the **scan rate** (forward and backward) **influences the repassivation capability**

→ scan rate for further tests:

- **forward scan:** 0.3 mV s⁻¹
- **backward scan:** 1 mV s⁻¹

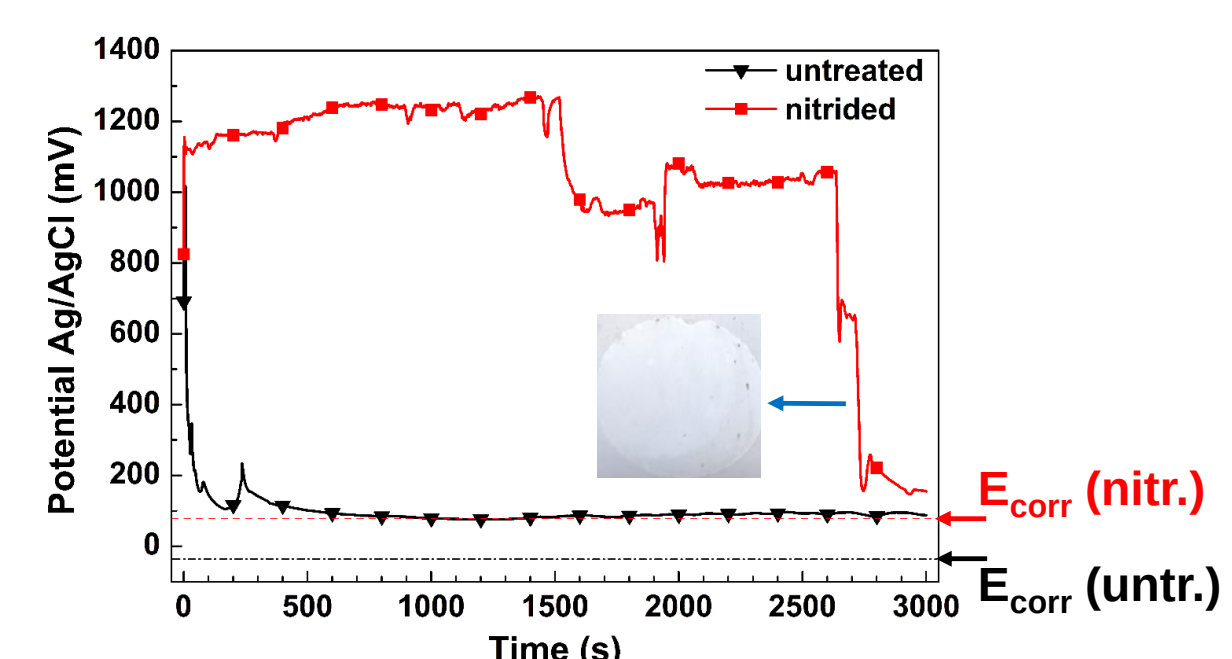


The **nitrided samples** have a **higher resistance to localized corrosion phenomena**, but they are not able to repassivate using these test conditions. Large colored regions are observed.



Galvanostatic tests

Potential values fluctuate due to transient passive film breakdown/repair. The **nitrided samples** have **potential values higher than the corrosion potential** of both the nitrided and untreated steel samples. **They could be able to repassivate**. Small pits are observed.



CONCLUSIONS

The **repassivation capability** of nitrided AISI 316L is particularly **sensitive to the extent of damage**. When **minor damage** happens, from, for example, using the galvanostatic technique, **high corrosion resistance** is maintained and **repassivation could occur**. However, fairly **significant corrosion damage**, such as that produced in cyclic potentiodynamic tests, **hinders repassivation**.